



Toward 3D Solid-State Batteries *via* **Atomic Layer Deposition Approach**

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3D solid-state batteries are receiving great attentions as on-board power supply systems for small-dimension devices, due to their high power and energy densities. However, the fabrication of 3D solid-state batteries has been a formidable challenge due to the limitation of conventional thin-film techniques. Recently, atomic layer deposition (ALD) has emerged as a powerful approach toward 3D solid-state batteries, because of its exclusive advantage of coating uniform, pinhole-free, and conformal functional thin films on high-aspect-ratio substrates. Herein, we review the most recent progress in the utilization of ALD for fabricating 3D solid-state batteries. Specifically, two aspects will be highlighted: the development of glassy solid-state batteries in half-cell and full-cell configurations *via* ALD approach. Based on this, the perspectives for further research will be discussed.

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INTRODUCTION

Solid-state thin-film batteries have been considered as promising technologies for powering a wide variety of small-dimension devices, such as microelectronics, medical implants, smart cards, and radio-frequency identification tags, due to their intrinsic safety and great flexibility in device design and integration (Long et al., 2004; Roberts et al., 2011). Early research have been focused on 2D solid-state batteries, which offered high power density owing to the fast Li ion and electron transport in thin films (Bates et al., 2000). However, 2D solid-state batteries have limitation in meeting growing demand for a higher energy density (mAh cm⁻²) from small-dimension devices. One approach to increase the energy density of 2D solid-state batteries is to increase the thickness of electrode thin films, which, however, will compromise battery power density because of the increased diffusion length for Li ions and electrons. One solution is switching from 2D to 3D solid-state batteries, which are structured on 3D substrates with high aspect ratio instead of planar substrates. 3D solid-state batteries will vastly improve both the energy and power densities, due to the significantly increased electrode area and maintained small thin-film thickness to insure fast Li ion and electron transport (Long et al., 2004; Oudenhoven et al., 2011; Roberts et al., 2011). Therefore, 3D solid-state batteries have been widely studied as the next-generation thin-film batteries in the past years.

Although several 3D solid-state battery configurations have been proposed, the fabrication of 3D solid-state batteries is challenging, due to the limitation of conventional thin-film techniques (such as physical vapor deposition, sputtering) (Oudenhoven et al., 2011; Roberts et al., 2011). 3D solid-state batteries require the step-wise deposition of conformal and pinhole-free functional thin films, including conductive layers, cathode, solid-state electrolyte (SSE), and anode, into 3D substrates. Atomic layer deposition (ALD) has emerged as a promising technique for 3D battery

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fabrication, because it can deposit high-quality thin films into 3D substrates. ALD utilizes a unique saturated and self-limiting reaction mechanism, which allows the deposition of a monolayer per ALD cycle (George, 2010). More importantly, ALD has been proven very effective in realizing conformal films in high-aspectratio substrates (Cheah et al., 2009; Liu et al., 2013). Therefore, enormous efforts have been devoted to developing the anode, cathode, and SSE materials by ALD, which are functional components for building 3D solid-state batteries. These materials, along with their ALD chemistry and processes have been summarized in several recent review papers (Knoops et al., 2012; Liu and Sun, 2015; Lu et al., 2017). This Mini Review is intended to be a highlight of recent breakthroughs in key material development by ALD and their applications in the integration of 3D solid-state battery devices. It is expected that this Review will shine lights on the future directions of material development and device design for 3D solid-state microbatteries by the ALD approach.

SSEs BY ALD

Fabrication of 3D batteries first requires the establishment of feasible ALD chemistry and processes for the anode, SSE, and cathode. Many anode (such as TiO₂, SnO₂, Co₃O₄, and GaS) and cathode (such as V₂O₅, LiCoO₂, LiFePO₄, and LiMnO₂) materials have been developed by ALD, and demonstrated good electrochemical performance in Li-ion batteries (Liu and Sun, 2015; Lu et al., 2017). Nevertheless, these anode and cathode are yet to be integrated into 3D solid-state batteries, and the lack of suitable SSEs by ALD becomes a limiting factor.

For the SSEs fabricated by ALD, they should possess the properties as required for all SSEs, such as high ionic conductivity, low electronic conductivity, good chemical stability, wide electrochemical window, and free from pine holes (Long et al., 2004). In addition, the SSEs by ALD need meet the following criteria in order to be used in 3D solid-state batteries. (1) Functional at the as-deposited state and without any harsh post treatment in order to preserve the continuity of the thin films (Liu et al., 2013; Kazyak et al., 2017). (2) ALD process should avoid complicated surface chemistry. By applying these criteria, the options for the SSEs by ALD are narrowed down to glassy inorganic SSEs (Liu et al., 2013; Cao et al., 2014).

Many glassy inorganic SSEs, such as LiAlO_x, Li₃PO₄, Li₂SiO₃, Li_{5.1}TaO_z, Li_xAl_yS, and LiAlF₄, have been deposited by ALD using

a "sub-cycle" strategy (Liu and Sun, 2015; Cao et al., 2016; Xie et al., 2017). The sub-cycle strategy combines two or more ALD sub-cycles, and is thus advantageous for controlling the composition of SSEs by adjusting the ratio between different ALD sub-cycles. However, it also introduces relatively complex ALD chemistry and processes, which could add additional complication to the fabrication of 3D solid-state batteries. Therefore, SSEs with simpler ALD process and higher ionic conductivity are in great need.

Recently, exciting breakthrough has been made by several groups to develop ALD processes for lithium phosphorus oxynitride (LiPON), which is a highly ion-conducting glassy electrolyte widely adopted in 2D thin-film batteries (Cao et al., 2014). LiPON was deposited in a temperature range of 200-500°C by using several ALD processes, such as [lithium tert-butoxide $(LiO^{t}Bu) - H_{2}O] + (trimethyl phosphate - plasma N_{2}) (Kozen)$ et al., 2015; Put et al., 2017) and (trisdimethylaminophospho $rus - O_2$ + (LiO^tBu - NH₃) (Shibata, 2016). Despite of the subcycle strategy employed, LiPON reported in these work showed a high ionic conductivity of $\sim 10^{-7}$ S cm⁻¹ at RT. By contrast, another work successfully deposited LiPON at 270-310°C by using just two precursors, lithium bis(trimethylsilyl)amide (LiHMDS) and diethyl phosphoramidate (DEPA) (Nisula et al., 2015). The growth rate of LiPON is ~0.7 Å/cycle, and the ionic conductivity of the LiPON deposited at 330°C reach as high as 6.6×10^{-7} S cm⁻¹ at 25°C, highest one reported so far (Nisula et al., 2015). Later on, it is found that the combination of LiO^tBu and DEPA also yielded self-limiting deposition of LiPON at temperatures of 250-300°C (Pearse et al., 2017a,b). Moreover, electrochemical analysis showed the good electrochemical stability of ALD-LiPON in a potential window of 0-5.3 V (vs. Li/Li⁺). More importantly, the LiPON thin film exhibited excellent coverage and uniformity in 3D Si trend substrates (Pearse et al., 2017a,b). Furthermore, the ALD-LiPON has been validated to be functional in 2D solid-state batteries (Nisula and Karppinen, 2016; Pearse et al., 2017a,b; Put et al., 2017). For example, 2D solid-state batteries has been fabricated by using sputtered LiCoO₂ as the cathode, ALD-LiPON as the electrolyte, and electron-beam-evaporated Si as the anode (Pearse et al., 2017a,b). The 2D LiCoO₂/LiPON/Si battery exhibited a cell capacity of ~16 μ Ah cm⁻² at a current of 300 µAh cm⁻² for 150 cycles (Table 1). Therefore, ALD-LiPON by ALD (LiHMDS or LiO'Bu-DEPA) holds great promise for

Method	Туре	Battery configuration ^a	Substrate	Electrolyte	Areal capacity (electrode dimension)	Reference
ALD	Hall cell	CNTs/FePO4/Li3PO4 Pt/TiO2/Li3PO4 Li4Ti5O12/LiPON/Li SS/Li2C8H4O4/LiPON	3D CNTs 3D Si 2D SiO ₂ 2D stainless steel (SS)	Li₃PO₄+liquid electrolyte Li₃PO₄+liquid electrolyte LiPON LiPON+liquid electrolyte	~40 μAh cm ⁻² (0.25 cm ²) ~370 μAh cm ⁻² (0.785 cm ²) ~0.3 Ah cm ⁻³ ~7 μAh cm ⁻²	Liu et al., 2016 Létiche et al., 2016 Put et al., 2017 Nisula and Karppinen, 2016
	Full cell	TiN _* /SnN _* /LiPON/LiV ₂ O ₅ /Ru Cu/Si/Li ₂ PO ₂ N/LiV ₂ O ₅ Cu/Si/Li ₂ PO ₂ N/LiCoO ₂ /Pt	3D Si 2D Si 2D Si	LiPON Li2PO2N Li2PO2N	~22 µAh cm ⁻² ~1.6 µAh cm ⁻² ~16 µAh cm ⁻²	Pearse et al., 2017a,b Pearse et al., 2017a,b
Sputtering	Hall cell	Li/LiPON/LiCoO2 ^b	2D planar	Lipon	~65 µAh cm ⁻²	Dudney, 2005

^aBattery configuration is writen in the order of conductive layer/anode/SSE/cathode/conductive layer, if applicable. ^bPlanar Li/LiPON/LiCoO₂ fabricated by Sputtering is provided as a benchmark. 3D solid-state battery application, considering its good ionic conductivity and straightforward ALD process.

3D SOLID-STATE BATTERIES BY ALD

Although ALD has been widely used to design 3D structured anode or cathode in Li-ion batteries (Liu and Sun, 2015; Lu et al., 2017), there have been few reports about using ALD to fabricate 3D solid-state batteries due to the lack of suitable SSEs. With the recent success in glassy SSEs by ALD, several recent works have demonstrated the integration of 3D solid-state batteries in half- or full-cell configurations by ALD, as illustrated in **Figure 1**.

In a half-cell 3D configuration, one electrode material was first deposited on 3D substrates, and then coated with SSEs by ALD. Compared with only 3D electrodes (Cheah et al., 2009), the integration of SSEs on the 3D electrodes by ALD is one big step toward 3D solid-state batteries. Even though the testing of half-cell 3D electrodes with SSEs is still performed in liquid electrolytes, the functionality of SSEs could be confirmed in 3D configuration (Létiche et al., 2016; Liu et al., 2016). For example, Liu et al. had grown carbon nanotubes (CNTs) on carbon fibers as both the 3D substrate and current collector. Amorphous FePO4 was deposited on the 3D CNTs by ALD, forming a 3D hierarchical CNTs@FePO4 architecture, which was then coated with Li₃PO₄ SSE by ALD at 250°C (Figure 1A) (Liu et al., 2016). The CNTs@FePO₄@Li₃PO₄ electrode (surface area of 0.25 cm²) exhibited an areal capacity of \sim 40 μ Ah cm⁻², 20 times higher than that on a planar substrate, in a liquid electrolyte (1 M LiPF₆ in EC:DEC:EMC). The areal capacity could be controlled by adjusting ALD cycles, which are linearly dependent with FePO₄ film thickness and loading. In another work, Manon et al. designed a 3D silicon wafer as a substrate with Pt as the current collector (Létiche et al., 2016). The half-cell solid-state battery (surface area 0.18 cm²) was integrated by step-wise deposition TiO₂ and Li₃PO₄ SSE by ALD (Figure 1B). The 3D TiO₂ electrodes with and without Li₃PO₄ SSE showed typical redox peaks at 1.7 and



FIGURE 1 | (A) Schematics of 3D CNTs@FePO4@Li₃PO4 half-cell battery fabricated by atomic layer deposition (ALD); (B) transmission X-ray microscope image of a similar micropillar coated with Al₂O₃/Pt/TiO₂/Li₃PO4 half-cell battery by ALD. Reproduced with the permission from John Wiley and Sons (Létiche et al., 2016; Liu et al., 2016). (C) The fabrication of 3D TiN/SnN₂/LiPON/LiV₂O₅/Ru full-cell battery on a 3D Si substrate by ALD. Reproduced with the permission from the authors (Pearse et al., 2017a,b).

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2.1 V in a liquid electrolyte (1 M LiTFSI in EC:DEC), indicating the ionic conduction nature of Li₃PO₄. The 3D TiO₂ (55 nm)/ Li₃PO₄ exhibited an areal capacity of 100 μ Ah cm⁻² at 1/10°C. When the thickness of TiO₂ increased to 155 nm, the areal capacity was enhanced to 370 μ Ah cm⁻², which is ~100 times higher than planar TiO₂ electrode (**Table 1**). These work proven that ALD is indeed an effective approach to integrate SSE onto the anode or Cathode. The areal capacity of the 3D battery can be easily controlled by ALD cycles, which have linear relation with the thickness and loading of active materials. Nevertheless, significant technical challenges remain in the ALD deposition and electrochemical testing of full 3D solid-state batteries.

A full-cell 3D solid-state battery has been fabricated by ALD for the first time recently (Pearse et al., 2017a,b). In this work, the 3D solid-state battery was patterned on a 3D Si substrate by sequentially depositing 40 nm Ru (cathode current collector), 70 nm prelithiated V₂O₅ (cathode), 50 nm LiPON (SSE), 10 nm SnN_x (anode), and 25 nm TiN (anode current collector), finally covered in a layer of electron-beam evaporated Cu (Figure 1C). It should be pointed out that all the active components of the 3D solid-state battery were deposited by ALD in this work. During the first cycle, the 3D battery on the substrates with 10 and 4 aspect ratios delivered a capacity of \sim 32 and 12 μ Ah cm⁻², respectively, which was 10.8 and 4.5 times relatively to the planar reference cell. Despite of irrversible capacity in the first few cycles, the 3D solid-state battery delivered a discharge capacity of ~22 µAh cm^{-2} at 100 cycles at 100 $\mu A~cm^{-2}$ (Table 1). This work is the first experimental proof of full-cell 3D solid-state battery realized by ALD, suggesting the feability and great potential of ALD for 3D battery device fabrication. Moreover, it is found that 40 nm LiPON was able to electrically isolate the anode and cathode in the 3D structure, indicating the complete coverage of ALD thin films on the 3D substrate. This finding is consistent with previous work in ALD-Li_{5.1}TaO_z, in which the SSE was also found to be functional in ~50 nm thickness (Liu et al., 2013). This is desirable for application because the reduced thickness of SSEs can decrease the cell impedance and ALD deposition time.

Table 1 lists both 2D and 3D solid-state batteries fabricated by ALD and their areal capacities, in comparison to a 2D thinfilm battery (Li/LiPON/LiCoO₂) fabricated by physical vapor deposition (Dudney, 2005). It is obvious that the areal capacity of 3D full-cell battery by ALD needs significant improvement to outperform the conventional Li/LiPON/LiCoO₂ 2D thin-film battery (Dudney, 2005). Further capacity improvement can be achieved by using higher-asepct-ratio substrates, increasing elecrode film thickness, and optimizing the system (Létiche et al., 2016; Liu et al., 2016; Pearse et al., 2017a,b).

SUMMARY AND PROSPECTIVE

In this Mini Review, we discussed recent progress in the deposition of SSEs and the fabrication of 3D solid-state batteries *via* ALD approach. With success in ALD chemistry and process for LiPON and Li₃PO₄, 3D solid-state batteries have been fabricated in half-cell or full-cell configurations by ALD. These works demonstrated the feasibility and potential of ALD technique in fabricating 3D solid-state batteries in a step-wise and wellcontrolled manner.

Regardless, further challenges remain in 3D solid-state batteries by ALD. First, the areal capacity of 3D solid-state batteries is to be further improved in order to outperform state-of-the-art 2D thin-film batteries. Future research should focus on using 3D substrates with higher aspect ratio, and thicker electrode thin films to enhance the areal capacity. Conformal deposition of thin films in high aspect ratio substrates (aspect ratio in excess of 1,000) requires much longer exposure and purging time for precursors in order to achieve complete surface reactions and byproduct removals, which would significantly increase the ALD processing time (Elam et al., 2003). The ALD processing time could be further prolonged when thicker electrod thin films, in other words more ALD cycles, are required. Roll-toroll ALD or spatial ALD is a potential solution ro reduce the ALD processing time and improve deposition efficiency (Poodt et al., 2010). Second, simpler ALD process and chemistry, as for ALD-LiPON and Li₃PO₄, is required for Li-containing cathode, which contains three or more elements. The cathode materials developed by ALD so far involved complex surface chemisty, which is unsuitable for 3D battery integration from practical point of view. A fundamental obstacle is that most available ALD precursors could only deliver one metal element (such as oxides and sulfides) in one ALD cycle. As a result, more ALD precursors, cycles, and surface chemistry are needed to achieve mutil-element compounds. Advacement relies on the development of novel ALD precursors, such as double metal alkoxides (Mäntymäki et al., 2012), which could lead to two metal elements in one ALD cycles. The ALD-LiPON (LiO'Bu-DEPA) is a good example of simplifying ALD surface chemisty, for which DEPA serves as both oxygen and nitrogen precursors. Third, the packaging and testing of 3D solid-state batteries needs more attentions. Although the recent work has reported a successful example (Pearse et al., 2017a,b), it requires multiple techniques in realizing the substrate conductive layers, and active materials. Finally yet importantly, the mechanical and chemical compatibilities of SSE with electrode thin films require further research attentions (Richards et al., 2016; Kerman et al., 2017). For example, stress/strain could develop in the electrode due to lithium intercalation/de-intercalation. Nevertheless, ALD has shown great promise in fabricating real 3D solid-state batteries on high-aspect-ratio substrates. It is expected that ALD will play a significant role in the development of practical 3D solid-state batteries for various applications, as such flexible electronics, MEMS.

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All authors discussed the topics and contributed to the organization of this paper.

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