



# Trap-Filling of ZnO Buffer Layer for Improved Efficiencies of Organic Solar Cells

# Mingguang Li<sup>1</sup>, Jing Li<sup>1</sup>, Longsheng Yu<sup>1</sup>, Ying Zhang<sup>1</sup>, Yizhong Dai<sup>1</sup>, Runfeng Chen<sup>1\*</sup> and Wei Huang<sup>1,2</sup>

<sup>1</sup> Key Laboratory for Organic Electronics and Information Displays & Jiangsu Key Laboratory for Biosensors, Institute of Advanced Materials (IAM), Jiangsu National Synergetic Innovation Center for Advanced Materials (SICAM), Nanjing University of Posts & Telecommunications, Nanjing, China, <sup>2</sup> Institute of Flexible Electronics, Northwestern Polytechnical University, Xi'an, China

Trap-assisted recombination loss in the cathode buffer layers (CBLs) is detrimental to the electron extraction process and severely restricts the power conversion efficiencies (PCEs) of organic solar cells (OSCs). Herein, a novel organic–inorganic hybrid film composed of zinc oxide (ZnO) and 2,3,5,6-tetrafluoro-7,7,8, 8-tetracyanoquinodimethane (F4TCNQ) is designed to fill the intrinsic charge traps of ZnO-based CBLs by doping F4TCNQ for high-performance inverted OSCs. Thus, constructed ZnO:F4TCNQ hybrid film exhibits enhanced surface hydrophobicity and adjustable energy levels, providing favorable interfacial condition for electron extraction process. Consequently, trap-assisted recombination loss in the CBLs was efficiently suppressed, leading to the significantly improved fill factor and PCEs of both fullerene- and non-fullerene-based OSCs using the ZnO:F4TCNQ hybrid CBLs. This work illustrates a convenient organic acceptor doping approach to suppress the internal charge traps of traditional inorganic CBLs, which will shed new light on the fabrication of high-performance CBLs with facile electron extraction processes in inverted OSC devices.

**OPEN ACCESS** 

#### Edited by:

Ye Zhou, Shenzhen University, China

#### Reviewed by:

Tao Liu, Hong Kong University of Science and Technology, Hong Kong Zong-Xiang Xu, Southern University of Science and Technology, China

#### \*Correspondence:

Runfeng Chen iamrfchen@njupt.edu.cn

#### Specialty section:

This article was submitted to Physical Chemistry and Chemical Physics, a section of the journal Frontiers in Chemistry

> Received: 09 January 2020 Accepted: 15 April 2020 Published: 26 May 2020

#### Citation:

Li M, Li J, Yu L, Zhang Y, Dai Y, Chen R and Huang W (2020) Trap-Filling of ZnO Buffer Layer for Improved Efficiencies of Organic Solar Cells. Front. Chem. 8:399. doi: 10.3389/fchem.2020.00399 Keywords: ZnO, cathode buffer layers, organic solar cells, trap-filling, organic-inorganic hybrid film

# INTRODUCTION

Bulk heterojunction (BHJ) organic solar cells (OSCs) have attracted extensive attention due to their intrinsic merits of light weight, high mechanical flexibility, and easy processing (Li et al., 2012; Yan et al., 2018). In the construction of high-performance OSC devices, both the regulation of photoactive layers and interface engineering of the integrated functional layers plays important roles in achieving high power conversion efficiencies (PCEs) (Liu et al., 2016, 2018; Liu T. et al., 2017). As for a specified photoactive layer, interfacial modification by using an efficient cathode buffer layer (CBL) has been widely applied to further optimize the device structure and improve the device efficiency (Yin et al., 2016).

Among various interfacial materials, the zinc oxide (ZnO) film inserted between organic photoactive layer and inorganic electrode is one of the most widely used CBLs in the inverted OSCs, exhibiting many inherent advantages, including high electron mobility, high optical transparency, tuneable electronic properties, ease of preparation, and low toxicity (Hewlett and McLachlan, 2016). Moreover, a simple and extensively investigated sol-gel method can be applied to fabricate

1

ZnO-based CBLs at a relatively mild condition (Kamalasanan and Chandra, 1996). However, the low-temperature solutionprocessed ZnO films generally have high density of surface and bulk defects, which would introduce undesired intra-gap energy levels and thus suppress carrier transport processes in the CBLs. These intrinsic defects in ZnO lattice mainly comprise oxygen vacancies (V<sub>O</sub>), oxygen interstitials (O<sub>i</sub>), zinc vacancies (V<sub>Zn</sub>), zinc interstitials (Zn<sub>i</sub>), etc. (Spencer et al., 2012). Plenty of defects acting as charge recombination centers will reduce the charge collection efficiency, which is an obstacle to the application of ZnO material in the field of optoelectronic devices (Sun et al., 2011; Lee et al., 2019).

In view of the above adverse issues, there have been a general consensus to optimize charge transporting properties of ZnO-based CBLs by passivating the defect sites (Hu et al., 2015; Cai et al., 2017; Li et al., 2017; Wang et al., 2019). For example, a high-performance fullerene-free OSC with efficiency over 12% was reported by employing ethylene diamine tetraacetic acid (EDTA)/ZnO hybrid interlayer. The introduced EDTA efficiently passivizes defects in ZnO film due to its chelation function, leading to balanced hole and electron mobility (Li et al., 2017). Besides, amphiphilic fullerenes/ZnO hybrids were also used as novel CBLs to passivate the defects of ZnO and the corresponding efficiency of the OSC based on thieno[3,4b]-thiophene/benzodithiophene (PTB7):PC71BM was up to 8.0% (Hu et al., 2015). Hybrid CBL consisting of ZnO and poly(2ethyl-2-oxazoline) (PEOz) was also constructed and applied to the inverted non-fullerene OSCs. Due to the reduced oxygen deficiency of ZnO by the introduced PEOz, an improved PCE over 12% was achieved (Seo et al., 2018). Additionally, various other organic materials such as polyethylenimine (PEI) (Liu C. et al., 2017), fullerene derivatives (Liao et al., 2014; Hu et al., 2016), and non-fullerene acceptors (Xie and Wuerthner, 2017; Li et al., 2019) have also been utilized to construct highperformance organic-inorganic hybrid materials by passivating the inherent defects of metal oxide layers and remarkable efficiency improvements were all observed.

The reported organic defect-passivating agents can generally coordinate with ZnO due to the existence of various polar groups, such as hydroxyl groups, amino groups, carbonyl groups, and carboxylic groups. Here, we found that 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodimethane (F4TCNQ) with four cyano (C=N) groups can also regulate the intrinsic property of ZnO CBLs with the assistance of coordination effects of C=N groups. In order to passivate the inherent defects originated from ZnObased CBLs, we propose an organic-inorganic hybrid CBL composed of inorganic ZnO and organic dopant F4TCNQ and explore how the F4TCNQ doping in the CBLs affects device operation. The introduction of ZnO:F4TCNQ hybrid CBLs is able to improve the PCEs of both fullerene and non-fullerene OSCs by the remarkable increase of fill factor (FF) value. The use of F4TCNQ dopant can modify surface properties and regulate energy-level structure of ZnO-based CBLs simultaneously. Benefiting from the coordination effect of C=N groups, trap density in the ZnO film is reduced significantly, which suppresses non-geminate charge recombination in the ZnO:F4TCNQ-based OSCs. These results indicate that the construction of novel ZnO:F4TCNQ hybrid film is a promising approach in the interfacial modification of ZnO-based CBLs for the performance optimization of OSC devices.

# **RESULTS AND DISCUSSION**

# **Photovoltaic Performance**

To passivate the ZnO defects such as oxygen vacancies and zinc dangling bonds and modify interfacial contact between metal electrode and organic photoactive layer, a novel organic-inorganic hybrid material based on ZnO and F4TCNQ was used as CBLs in both fullerene- and nonfullerene-based OSCs. **Figure 1A** provides the molecular structure of photoactive materials: donor poly[4,8-bis(5-(2ethylhexyl)thiophen-2-yl)benzo[1,2-b:4,5-b']dithiophene-

co-fluorothieno[3,4-b]thiophene-2-carboxylate] (PTB7-Th), fullerene-based acceptor  $PC_{71}BM$ , and non-fullerene-based acceptor 3,9-bis(2-methylene-(3-(1,1-dicyanomethylene)indanone))-5,5,11,11-tetrakis(4-hexylphenyl)-dithieno[2,3-

d:2',3'-d<sup>'</sup>]-s-indaceno[1,2-b:5,6-b<sup>'</sup>]dithiophene) (ITIC). The molecular structure of organic dopant F4TCNQ is also supplied in **Figure 1A**. Meanwhile, **Figure 1B** shows the schematic device structure and the corresponding energy levels. Considering that sol-gel-derived ZnO generally has high density of defects, the organic dopant F4TCNQ was employed into the precursor solution to passivate the defects in both surface and bulk. Similar to the previous approach used to fabricate pure ZnO film, ZnO:F4TCNQ hybrid films with different doping concentrations were deposited by an identical sol-gel strategy (Sun et al., 2011; Li et al., 2019).

As shown in Figure 2A, J-V characteristics of the inversed OSCs based on the PTB7-Th:PC71BM system using ZnO:F4TCNQ hybrid CBLs are investigated under AM 1.5G irradiation. The corresponding photovoltaic parameters including open-circuit voltage ( $V_{OC}$ ), short-circuit current density  $(J_{SC})$ , FF, PCE, series resistance  $(R_S)$ , and shunt resistance  $(R_{\rm sh})$  are summarized in Table 1. The inverted OSC device with the structure of indium tin oxide (ITO)/CBL/PTB7-Th:PC<sub>71</sub>BM/MoO<sub>3</sub>/Al is shown in Figure 2B. For the devices fabricated with ZnO:F4TCNQ hybrid CBLs, the PCE value enhances from 7.17% for pristine device to 7.55% for ZnO:0.5 wt% F4TCNQ device, 8.14% for ZnO:1.0 wt% F4TCNQ device, and 7.56% for ZnO:5.0 wt% F4TCNQ device, respectively. When organic dopant F4TCNQ with the optimal concentration of 1.0 wt% is introduced into ZnO film, the  $J_{SC}$  value increases slightly from 14.24 to 15.00 mA/cm<sup>2</sup> while the FF improved from 62.2% to 67.1% significantly. The slightly enhanced external quantum efficiency (EQE) almost in the entire absorption range shown in Figure 2B is consistent with the minor variations of  $J_{\rm SC}$  value as well as the slight reduction of  $R_{\rm S}$ . Furthermore, the ZnO:F4TCNQ hybrid thin film used as a novel CBL was also applied to PTB7-Th:ITIC-based non-fullerene OSCs. Since the LUMO levels of PC71BM and ITIC are quite close, the VOC values are the same for both fullerene- and nonfullerene-based OSCs. The non-fullerene-based device exhibits improved PCE values from 6.13 to 6.96%, which also mainly contributed to the enhancement of FF values (Table 1 and



**Supplementary Figure 1**). Therefore, the construction of ZnO:F4TCNQ hybrid CBLs could efficiently improve the final device performance of both fullerene and non-fullerene OSCs by enhancing the FF values.

The J-V characteristics of inverted OSCs fabricated with ZnO:F4TCNQ hybrid CBLs in the dark condition were measured

to investigate the inherent electrical properties, as shown in **Figure 2C**. Compared to the pristine ZnO-based device, the introduction of 1.0 wt% F4TCNQ leads to a significant reduction of leakage current, which demonstrates the efficient hole blocking capability of the ZnO:F4TCNQ hybrid film. On the contrary, the leakage current becomes strengthened with the



further increase of F4TCNQ content (5 wt%). To investigate the charge recombination dynamics in the PTB7-Th:PC71BM system, the dependence of  $J_{SC}$  on light intensity ( $P_{light}$ ) was also conducted under the optimal F4TCNQ dopant content (Figure 2D). Generally, the dependence of  $J_{SC}$  on  $P_{light}$  can be expressed as  $J_{SC} \propto (P_{\text{light}})^{\alpha}$  (Zhao et al., 2017). Figure 2D shows that the fitted slope ( $\alpha = 0.897$ ) of the ZnO:F4TCNQbased device is much closer to 1, thus demonstrating the beneficial effect of the introduced F4TCNQ dopant on the reduction of bimolecular recombination process. In addition, the significantly enhanced  $R_{\rm sh}$  from 508.2 to 913.8  $\Omega \cdot \rm{cm}^2$  for the ZnO:F4TCNQ-based device also indicates a weakened carrier recombination at the ZnO/photoactive layer interfaces (Yin et al., 2013). Therefore, the significantly enlarged  $R_{\rm sh}$  and the restrained leakage current suggest that the introduced F4TCNQ component plays a significant role in optimizing the interfacial contact between different functional layers, thus facilitating the remarkable enhancement of FF value. When F4TCNQ was doped into ZnO film, a similar R<sub>sh</sub> enhanced rule is observed for nonfullerene OSCs. Consequently, organic dopant F4TCNQ in both fullerene and non-fullerene systems is capable of suppressing charge recombination events, thus significantly enhancing the FF and PCE values of OSCs.

#### Properties of ZnO:F4TCNQ Hybrid Thin Films

X-ray photoelectron spectroscopy (XPS) measurements were conducted to explore the effect of F4TCNQ dopant on the formation of ZnO-based CBLs (**Supplementary Figure 2**). The high-resolution XPS spectra of Zn  $2p_{1/2}$ , Zn  $2p_{3/2}$ , O 1s, and N 1s are shown in **Figures 3A–C**. When 1.0 wt% F4TCNQ was doped, the peaks of Zn  $2p_{1/2}$  (~1043.0 eV) and Zn  $2p_{3/2}$  (~1019.9 eV) shift toward lower binding energy by ~0.2 and ~0.1 eV, respectively. Meanwhile, the main peak of O 1s (~528.8 eV) corresponding to O atoms in ZnO matrix also shifts toward lower binding energy by ~0.1 eV. The peak shifts shown in **Figures 3A,B** indicate that more Zn atoms are bound to O atoms and the oxygen-deficient defects originated from hydroxyl groups or carboxylate groups are passivated in the ZnO:F4TCNQ hybrid film (Sun et al., 2011; Li et al., 2017). In

System	CBL	V <sub>oc</sub> (V)	J <sub>sc</sub> (mA/cm²)	FF (%)	PCE (%)	R <sub>s</sub> <sup>a</sup> (Ω ⋅cm²)	R <sub>sh</sub> <sup>b</sup> (Ω⋅cm²)
0.5 wt% F4TCNQ	0.81	14.50	66.0	7.75	5.26	662.1	
1.0 wt% F4TCNQ	0.81	15.00	67.1	8.14	4.41	913.8	
5.0 wt% F4TCNQ	0.81	14.03	66.6	7.56	6.62	839.8	
PTB7-Th:ITIC	ZnO	0.81	14.35	52.8	6.13	11.5	415.4
	0.5 wt% F4TCNQ	0.81	14.48	57.1	6.69	12.3	475.7
	1.0 wt% F4TCNQ	0.81	14.75	58.3	6.96	9.44	549.5
	5.0 wt% F4TCNQ	0.81	14.87	57.4	6.91	9.71	484.5

TABLE 1 | Performance of inverted fullerene- and non-fullerene-based OSCs with different CBLs.

<sup>a</sup>Series resistance ( $R_s$ ) and <sup>b</sup>Shunt resistance ( $R_{sh}$ ).

addition, compared to the pure ZnO film, the appearance of N 1s peak for ZnO:F4TCNQ hybrid film further demonstrates that the organic dopant F4TCNQ has been incorporated into ZnO film.

Cyclic voltammetry (CV) measurements combined with UVvis absorption spectra were performed to investigate the impact of introduced F4TCNQ on the energy levels of ZnO-based CBLs. As shown in Figure 3D, the employed F4TCNQ dopant exhibits a marginal impact on the optical properties of ZnO film and thus it can be approximately considered that the band gap of ZnO-based CBLs remains unchanged when a small amount of F4TCNQ molecules are added into ZnO precursor solution. Meanwhile, the highest occupied molecular orbital (HOMO) levels of ZnO and ZnO:F4TCNQ hybrid films were characterized by CV measurement (Figure 3E). The HOMO level of pure ZnO calculated from the onset potential of oxidation process is found to be -7.15 eV. While, the onset potential of ZnO:1.0 wt% F4TCNQ oxidation process shifts toward larger potential position and the corresponding HOMO level is calculated to be -7.29 eV. When the content of F4TCNQ is further raised to 5.0 wt%, the HOMO level of ZnO:F4TCNQ hybrid CBL reduces to -7.56 eV. According to the UV-vis absorption spectra and CV measurements of ZnO-based CBLs, the corresponding lowest unoccupied molecular orbital (LUMO) level can be obtained, as shown in the inset of Figure 3E. The optimized energy level structure by employing F4TCNQ dopant can lower the charge barrier of electron collection and thus be expected to facilitate the acquirement of high-performance OSCs.

In addition, we also performed atomic force microscopy (AFM) and contact angle ( $\theta$ ) measurement to reveal the influence of F4TCNQ dopant on the surface properties of ZnO-based CBLs, as shown in the inset of **Figure 3D**. Compared to the pure ZnO film with the roughness of 0.809 nm, the ZnO:1.0 wt% F4TCNQ hybrid film exhibits similar film roughness (0.961 nm). When the content of F4TCNQ reaches 5.0 wt%, the roughness is further increased to more than 1 nm and the remarkable aggregates can be observed at the film surface (**Supplementary Figure 3**). Therefore, the content of F4TCNQ dopant should be adjusted cautiously to fabricate the ZnO-based hybrid CBLs with excellent interfacial properties. Besides, the contact angle ( $\theta$ ) measurement provides the hydrophobic property of ZnO-based CBLs (the inset of

**Figure 3D**). Due to the excellent hydrophobicity of F4TCNQ itself, the constructed ZnO:F4TCNQ hybrid CBL exhibits an enhanced hydrophobic property with the  $\theta$  of 45.7°, which can perform a beneficial role in optimizing interfacial contact between CBL and organic photoactive layer (Hau et al., 2008; Liao et al., 2013). Furthermore, the conductivity measurements of the ZnO and ZnO:F4TCNQ films were also conducted (**Supplementary Table 1**). It was found that the ZnO:F4TCNQ film slightly decreases from  $2.06 \times 10^{-3}$  to  $1.56 \times 10^{-3}$  S/cm, indicating that the electron transfer from ZnO to the organic electron-acceptor molecule is not very efficient (Hewlett and McLachlan, 2016). This variation further demonstrates that the primary function of F4TCNQ is to optimize the interfacial contact rather than the inherent carrier transport properties of ZnO.

#### **Trap-Filling Mechanism**

Fourier transform infrared (FTIR) spectra are conducted to investigate the intermolecular interaction between F4TCNQ and ZnO (Figure 4A). In the pure F4TCNQ film, the stretching vibrations related to the four C=N groups appear at 2,224 and 2,178 cm<sup>-1</sup>, respectively (Kato et al., 1991), while there is no characteristic peak that appeared at around  $2,200 \text{ cm}^{-1}$  for pure ZnO film. As soon as 1 wt% F4TCNQ was doped into ZnO, the two characteristic peaks of C=N stretching vibration shift to 2,204 and 2,158 cm<sup>-1</sup>, respectively. Considering that the peak signal of C=N stretching vibration is relatively weak in the ZnO:1.0 wt% F4TCNQ film, the sample of ZnO:10.0 wt% F4TCNQ film was also detected and the same changes have been observed. The red shift of C=N stretching vibration peaks indicates that the C=N groups in F4TCNQ molecules coordinate with ZnO (Zhou et al., 2016). Benefiting from the coordination effects between F4TCNQ and ZnO, the passivation of various defects that existed in the ZnO matrix can be anticipated.

According to the previous reports, the photoluminescence (PL) properties of ZnO are very sensitive to the defects inside (Chen et al., 2012; Mishra et al., 2019). As shown in **Figure 4B**, the narrow emission peak at 363 nm is attributed to the band-toband emission of ZnO, while the broad emission one at around 600 nm stems from the defect states in ZnO matrix (Mishra et al., 2019). The broad emissions correspond to various defects,

N 1s

С

ntensity (a. u.)

526

ZnO

-4.46

F4TCNQ

5.0 wt%

-7.56

1.2

0.6

408

5.0 wt% F4TCNQ

-4.19

F4TCNQ

1.0 wt%

-7.29

1.8

404

- 1.0 wt% F4TCNQ

4.05

ZnO

-7.15

Voltage (V)

2.4

3.0

3.6

400

Binding energy (eV)

396

392

0 1s



в

Intensity (a. u.)

532

1.0 wt% F4TCNQ

RMS = 0.961 nm

 $\theta = 45.7^{\circ}$ 

700

1 µm

800

531 530 529 528 527

Е

Current (10<sup>-5</sup> A)

Binding energy (eV)

Zn 2p

2p<sub>3/2</sub>

1042 1022 1020 1018

ZnO

RMS = 0.809 nm

 $\theta = 27.2^{\circ}$ 

500

Wavelength (nm)

images (inset), (E) CV measurements and energy level diagram (inset) of various ZnO-based CBLs.

600

400

1 µm

Binding energy (eV)

states. On the contrary, when the content of F41CNQ is further increased to 5 wt%, the condition of defect states deteriorates and the emission peak corresponding to the defect states becomes even stronger. We speculate that excessive doping of F4TCNQ induces the formation of new defects in the ZnO-based CBLs. To exclude the impact of emission interference from F4TCNQ itself,

the PL spectrum of the pure F4TCNQ film was also measured and the corresponding fluorescence signal in the range of visible region was found to be very weak (**Supplementary Figure 4**). The results indicate that an appropriate amount of F4TCNQ dopant can conduce to the construction of an excellent ZnO thin film with less defect states. In addition, the time-resolved PL spectra of pure PC<sub>71</sub>BM layers were also measured and the exciton lifetime of PC<sub>71</sub>BM on the ITO/ZnO:F4TCNQ substrate was found to be slightly longer than that on the ITO/ZnO substrate. This confirms that the photo-generated excitons prefer to transfer to the ZnO:F4TCNQ layer, leading to reduced charge recombination (**Supplementary Figure 5**).

Trap-assisted recombination that happened in the ZnO film can be regarded as a two-step process (**Supplementary Figure 6**); i.e., a hole is first trapped within the band gap; then, an oppositely

Α

Intensity (a. u.)

D

Absorption Intensity (a.u.)

1.0

0.8

0.6

0.4

0.2

0.0

300

ZnO

1044

1.0 wt% F4TCNQ

2p<sub>1/2</sub>



charged electron finds the trapped hole and thus recombines with it (Proctor et al., 2013). Generally, the holes should be transported through the donor phase to the anode electrode

when the photogenerated carriers have formed in the photoactive layer. However, a mass of defects generated in the ZnO film may induce hole transfer from the donor component to the ZnO CBL. When the electrons are collected through ZnO-based CBLs, they may enter the trap states with positively charged species and trap-assisted recombination process happens. According to the previous reports, the trap-assisted recombination rate is ultimately determined by the amount of trap sites and how quickly the free carrier can find the trapped carrier (Kirchartz et al., 2011; Proctor et al., 2013). As soon as F4TCNQ dopants were employed, the introduction of F4TCNQ dopants will fill the defects where traps reside, thus rendering them electronically inert (Yan et al., 2016). Based on the trap-filling strategy, the photogenerated holes will be blocked from entering trap states and thus the probability of trap-assisted recombination can be reduced efficiently during carrier transport process.

### CONCLUSIONS

In conclusion, the ZnO:F4TCNQ hybrid films with good solution processability, tunable energy levels, and reduced defects have been developed as a new class of CBLs for both fullereneand non-fullerene-based OSCs. While adding F4TCNQ into ZnO film, remarkable increases in FF were observed, leading to an obvious enhancement in overall device performance. The surface property and energy levels of ZnO-based CBLs can be regulated efficiently by the doping of organic acceptor. Owing to the coordination effects between cyano groups and ZnO, the introduced F4TCNQ contributes to suppressing the nongeminate recombination by filling the ZnO defects, thus allowing for efficient charge transport and extraction in ZnO-based CBLs. Consequently, the simple trap-filling strategy by organic dopants provides an efficient approach in adjusting electrical properties of interfacial layers, which is important for constructing highperformance CBLs of inverted OSCs.

### **EXPERIMENTAL**

#### **Materials**

PTB7-Th, PC<sub>71</sub>BM, and ITIC were purchased from Luminescence Technology Corp. O-dichlorobenzene (ODCB) and chlorobenzene (CB) were purchased from Aladdin Corp. Zinc acetate dihydrate [Zn(CH<sub>3</sub>COO)<sub>2</sub>•2H<sub>2</sub>O, 98%], ethanolamine (NH<sub>2</sub>CH<sub>2</sub>CH<sub>2</sub>OH, 99.5%), and 2-methoxyethanol (CH<sub>3</sub>OCH<sub>2</sub>CH<sub>2</sub>OH, 99.8%) were purchased from Acros. 2,3,5,6-Tetrafluoro-7,7,8,8-tetracyanoquinodimethane (F4TCNQ) was provided by Sigma-Aldrich. Molybdenum oxide (MoO<sub>3</sub>) was provided by Shanghai Han Feng Chemical Corp. All chemicals were used as obtained.

### **Preparation of CBLs**

Sol-gel-derived ZnO CBLs were prepared using zinc acetate in 2-methoxyethanol:ethanolamine as a precursor solution by spincoating onto the ITO substrate followed by thermal annealing at 200°C for 30 min according to the reference (Sun et al., 2011). As for the ZnO:F4TCNQ hybrid CBLs, the ZnO:F4TCNQ precursor solutions were prepared by mixing with different contents of F4TCNQ (0.5, 1.0, and 5.0 wt%) and dissolving in a mixture of 2methoxyethanol as the solvent and ethanolamine as the stabilizer under sufficient stirring. After the spin-coating process has been completed, the hybrid film was also thermal-annealed at  $200^{\circ}$ C for 30 min.

#### **Device Fabrication**

The device structure was ITO/CBL/PTB7-Th:PC<sub>71</sub>BM/MoO<sub>3</sub>/Al. Firstly, the CBLs (~30 nm) were spin-coated on top of the precleaned ITO substrate, which was treated with deionized water, acetone, and ethanol in an ultrasonic bath followed by ultraviolet (UV)-ozone treating for 15 min. A blend of PTB7-Th:PC71 BM with the mass ratio of 4:7 in ODCB solvent was spin-coated on the surface of CBLs in the N<sub>2</sub> glovebox. The thickness of the active layer was controlled to be around 100 nm. Then, MoO3 film (10 nm) as the anode interfacial layer was thermally evaporated on the active layer, followed by the thermal evaporation of Al (~100 nm). The device of ITO/ZnO/PTB7-Th:ITIC/MoO3/Al was prepared by adopting a similar approach. For the non-fullerene-based photoactive layer, a blend of PTB7-Th:ITIC with the mass ratio of 1:1.2 in CB solvent was spin-coated. The thickness of the active layer was also controlled to be around 100 nm. These OSCs, made on ITO glass substrate, have an active area of  $0.09 \text{ cm}^2$ .

#### **Measurements and Characterization**

Current density-voltage (J-V) characteristics of OSCs, which were detected under ambient situation, were recorded by Keithley-2400 digital source meter. SAN-EI electric solar simulator (XES-50S1) under simulated AM 1.5 illumination  $(100 \text{ mW/cm}^2)$  was used for solar cell characterization. The EQE measurements of the PSCs were carried out with QE-R system (Enli Technology Co., Ltd.). The surface morphology of the CBLs was characterized by atomic force microscopy (AFM, FM-Nanoview 1000, FSM-Precision Co., Ltd.) in taping mode. Contact angle characterizations were carried out on pure ZnO layer, and ZnO:F4TCNQ layer using droplet shape analyzer (KRÜSS GmbH Germany). Xray photoelectron spectroscopy (XPS) experiments were carried out with ESCALAB 250 system. The measurement chamber was equipped with a monochromatic Al Ky X-ray source  $(h\gamma = 1486.6 \text{ eV})$ . Ultraviolet-visible (UV-Vis) absorption

# REFERENCES

- Cai, X., Yuan, T., Liu, X., and Tee, G. (2017). Self-assembly of 1-pyrenemethanol on ZnO surface toward combined cathode buffer layers for inverted polymer solar cells. *ACS Appl. Mater. Inter.* 9, 36082–36089. doi: 10.1021/acsami.7b10399
- Chen, S., Small, C. E., Amb, C. M., Subbiah, J., Lai, T. H., Tsang, S. W., et al. (2012). Inverted polymer solar cells with reduced interface recombination. *Adv. Energy Mater.* 2, 1333–1337. doi: 10.1002/aenm.201200184
- Hau, S. K., Yip, H. L., Ma, H., and Jen, A. K. Y. (2008). High performance ambient processed inverted polymer solar cells through interfacial modification with a fullerene self-assembled monolayer. *Appl. Phys. Lett.* 93:233304. doi: 10.1063/1.3028094

spectra were measured on a lambda 35 PerkinElmer UV-Vis spectrophotometer. Photoluminescence (PL) spectra and timeresolved PL measurements were performed on an Edinburgh FLS980 fluorescence at room temperature. CV measurement was performed to estimate the HOMO from the onset potential of the electrochemical oxidation. The CV measurement was carried out on a CHI660E system in a three-electrode cell with a working electrode, a reference electrode, and a counter electrode in an acetonitrile solution of  $Bu_4NPF_6$  (0.1 M). Fourier transform infrared (FTIR) spectroscopic analysis was performed using FT-IR spectrometer (PerkinElmer). The electrical conductivity was measured by Physical Property Measurement System (Quantum Design).

# DATA AVAILABILITY STATEMENT

All relevant data is contained within the article. The raw data supporting the conclusions of this article will be made available by the authors, without undue reservation, to any qualified researcher.

# **AUTHOR CONTRIBUTIONS**

ML and JL conceived the ideas and performed the device fabrication. LY and YZ conducted the XPS, UV-vis, and CV measurements. YD conducted the AFM and FTIR measurements. RC and WH supervised the research and contributed to manuscript preparation.

# FUNDING

This study was supported in part by the National Natural Science Foundation of China (61704089, 61875090, and 21674049), China Postdoctoral Science Foundation (2019M661899), Jiangsu Planned Projects for Postdoctoral Research Funds (2019K140), and Scientific Starting Fund from Nanjing University of Posts and Telecommunications (NUPTSF) (NY217141).

# SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: https://www.frontiersin.org/articles/10.3389/fchem. 2020.00399/full#supplementary-material

- Hewlett, R. M., and McLachlan, M. A. (2016). Surface structure modification of ZnO and the impact on electronic properties. *Adv. Mater.* 28, 3893–3921. doi: 10.1002/adma.201503404
- Hu, T., Chen, L., Yuan, K., and Chen, Y. W. (2015). Amphiphilic fullerene/ZnO hybrids as cathode buffer layers to improve charge selectivity of inverted polymer solar cells. *Nanoscale* 7, 9194–9203. doi: 10.1039/c5nr01456e
- Hu, T., Jiang, P., Chen, L., Yuan, K., Yang, H., and Chen, Y. (2016). Amphiphilic fullerene derivative as effective interfacial layer for inverted polymer solar cells. *Org. Electron.* 37, 35–41. doi: 10.1016/j.orgel.2016.06.018
- Kamalasanan, M. N., and Chandra, S. (1996). Sol-gel synthesis of ZnO thin films. *Thin Solid Films* 288, 112–115. doi: 10.1016/S0040-6090(96)0 8864-5

- Kato, H., Nishikawa, O., Matsui, T., Honma, S., and Kokado, H. (1991). Fouriertransform infrared-spectroscopy study of dopants of polypyrrole. J. Phys. Chem. 95, 6011–6014. doi: 10.1021/j100168a054
- Kirchartz, T., Pieters, B. E., Kirkpatrick, J., Rau, U., and Nelson, J. (2011). Recombination via tail states in polythiophene:fullerene solar cells. *Phys. Rev. B* 83:115209. doi: 10.1103/PhysRevB.83.115209
- Lee, S., Hang, C. Y., Hong, A., Kim, J., Yang, H., Jung, B. J., et al. (2019). Inverted quantum dot light-emitting diodes with defect-passivated ZnO as an electron transport layer. *Semicond. Sci. Tech.* 34:085002. doi: 10.1088/1361-6641/ab2b52
- Li, G., Zhu, R., and Yang, Y. (2012). Polymer solar cells. *Nat. Photonics* 6, 153–161. doi: 10.1038/NPHOTON.2012.11
- Li, M., Li, J., Wang, J., Yu, L., Wang, T., Jin, J., et al. (2019). Enhancing photovoltaic performance by cathode interfacial modification with inorganic/organic gradient diffusion structures. ACS Sustain. Chem. Eng. 7, 12261–12269. doi: 10.1021/acssuschemeng.9b01601
- Li, X., Liu, X., Zhang, W., Wang, H. Q., and Fang, J. (2017). Fullerene-free organic solar cells with efficiency over 12% based on EDTA-ZnO hybrid cathode interlayer. *Chem. Mater.* 29, 4176–4180. doi: 10.1021/acs.chemmater.7b01615
- Liao, S. H., Jhuo, H. J., Cheng, Y. S., and Chen, S. A. (2013). Fullerene derivativedoped zinc oxide nanofilm as the cathode of inverted polymer solar cells with low-bandgap polymer (PTB7-Th) for high performance. *Adv. Mater.* 25, 4766–4771. doi: 10.1002/adma.201301476
- Liao, S. H., Jhuo, H. J., Yeh, P. N., Cheng, Y. S., Li, Y. L., Lee, Y. H., et al. (2014). Single junction inverted polymer solar cell reaching power conversion efficiency 10.31% by employing dual-doped zinc oxide nano-film as cathode interlayer. *Sci. Rep.* 4:6813. doi: 10.1038/srep06813
- Liu, C., Li, Z., Zhang, X., Guo, W., Zhang, L., and Ruan, S. (2017). Annealingfree ZnO:PEI composite cathode interfacial layer for efficient organic solar cells. ACS Photonics 4, 2952–2958. doi: 10.1021/acsphotonics.7b01096
- Liu, T., Guo, Y., Yi, Y., Huo, L., Xue, X., Sun, X., et al. (2016). Ternary organic solar cells based on two compatible nonfullerene acceptors with power conversion efficiency > 10%. Adv. Mater. 28, 10008–10015. doi: 10.1002/adma.20160257
- Liu, T., Luo, Z., Fan, Q., Zhang, G., Zhang, L., Gao, W., et al. (2018). Use of two structurally similar small molecular acceptors enabling ternary organic solar cells with high efficiencies and fill factors. *Energ. Environ. Sci.* 11, 3275–3282. doi: 10.1039/C8EE01700J
- Liu, T., Xue, X., Huo, L., Sun, X., An, Q., Zhang, F., et al. (2017). Highly efficient parallel-like ternary organic solar cells. *Chem. Mater.* 29, 2914–2920. doi: 10.1021/acs.chemmater.6b05194
- Mishra, P. K., Ayaz, S., Srivastava, T., Tiwari, S., Meena, R., Kissinquinker, B., et al. (2019). Role of Ga-substitution in ZnO on defect states, carrier density, mobility and UV sensing. *J. Mater. Sci-Mater. Electron.* 30, 18686–18695. doi: 10.1007/s10854-019-02221-z
- Proctor, C. M., Kuik, M., and Nguyen, T. (2013). Charge carrier recombination in organic solar cells. *Prog. Polym. Sci.* 38, 1941–1960. doi: 10.1016/j.progpolymsci.2013.08.008
- Seo, J., Lee, C., Kim, H., and Kim, Y. (2018). Distinctive nanocrater structures in hybrid electron-collecting buffer layers for high efficiency polymer:nonfullerene solar cells. *Adv. Mater. Inter.* 5:1800912. doi: 10.1002/admi.201800912

- Spencer, M. J. S., Wong, K. W. J., and Yarovsky, I. (2012). Surface defects on ZnO nanowires: implications for design of sensors. J. Phys-Condens. Matter 24:305001. doi: 10.1088/0953-8984/24/30/305001
- Sun, Y., Seo, J. H., Takacs, C. J., Seifter, J., and Heeger, A. J. (2011). Inverted polymer solar cells integrated with a low-temperature-annealed sol-gelderived ZnO film as an electron transport layer. *Adv. Mater.* 23, 1679–1683. doi: 10.1002/adma.201004301
- Wang, C., Luo, D., Gao, Y., Wang, G., Wang, C., Ma, P., et al. (2019). Delicate energy-level adjustment and interfacial defect passivation of ZnO electron transport layers in organic solar cells by constructing ZnO/In nanojunctions. J. Phys. Chem. C 123, 16546–16555. doi: 10.1021/acs.jpcc.9b 03776
- Xie, Z., and Wuerthner, F. (2017). Hybrid photoconductive cathode interlayer materials composed of perylene bisimide photosensitizers and zinc oxide for high performance polymer solar cells. *Adv. Energy Mater.* 7:1602573. doi: 10.1002/aenm.201602573
- Yan, C., Barlow, S., Wang, Z., Yan, H., Jen, A. K. Y., Marder, S. R., et al. (2018). Non-fullerene acceptors for organic solar cells. *Nat. Rev. Mater.* 3: 18003. doi: 10.1038/natrevmats.2018.3
- Yan, H., Manion, J. G., Yuan, M., de Arquer, F. P. G., McKeown, G. R., Beaupre, S., et al. (2016). Increasing polymer solar cell fill factor by trap-filling with F4-TCNQ at parts per thousand concentration. *Adv. Mater.* 28, 6491–6496. doi: 10.1002/adma.201601553
- Yin, Z., Wei, J., and Zheng, Q. (2016). Interfacial materials for organic solar cells: recent advances and perspectives. *Adv. Sci.* 3:1500362. doi: 10.1002/advs.201500362
- Yin, Z., Zheng, Q., Chen, S. C., and Cai, D. (2013). Interface control of semiconducting metal oxide layers for efficient and stable inverted polymer solar cells with open-circuit voltages over 1.0 volt. ACS Appl. Mater. Inter. 5, 9015–9025. doi: 10.1021/am402175m
- Zhao, F., Li, Y., Wang, Z., Yang, Y., Wang, Z., He, G., et al. (2017). Combining energy transfer and optimized morphology for highly efficient ternary polymer solar cells. Adv. Energy Mater. 7: 1602552. doi: 10.1002/aenm.2016 02552
- Zhou, J., Fan, J. B., Nie, Q., and Wang, S. (2016). Three-dimensional superhydrophobic copper 7,7,8,8-tetracyanoquinodimethane biointerfaces with the capability of high adhesion of osteoblasts. *Nanoscale* 8, 3264–3267. doi: 10.1039/c5nr08305b

**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

Copyright © 2020 Li, Li, Yu, Zhang, Dai, Chen and Huang. This is an open-access article distributed under the terms of the Creative Commons Attribution License (CC BY). The use, distribution or reproduction in other forums is permitted, provided the original author(s) and the copyright owner(s) are credited and that the original publication in this journal is cited, in accordance with accepted academic practice. No use, distribution or reproduction is permitted which does not comply with these terms.