



# Performance Optimization for PbTe-Based Thermoelectric Materials

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Thermoelectric material is a kind of functional material that uses the movement of carriers inside a solid to realize the direct mutual conversion of heat and electric energy. It provides a safe, reliable, pollution-free, noise-free, all-solid-state power generation and cooling method and has a wide range of application prospects. Among them, the characteristics of low valence band degeneracy, low effective quality of conduction band, strong phonon non-harmonicity, simple crystal structure and adjustable microstructure have made PbTe-based materials the focus of research in the thermoelectric field. In this review, two latest strategies to improve the thermoelectric properties of PbTe-based materials are discussed, and the challenges for the further development of PbTe-based thermoelectric materials and the prospects for the future are also outlined.

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# INTRODUCTION

PbTe has a highly symmetric face-centered cubic crystal structure, as shown in Figure 1A (Xiao and Zhao, 2018), so it has excellent electrical conductivity. Its electronic energy band structure shows that the maximum value of the valence band is 7.00389 ev, the minimum value of the conduction band is 7.80069 ev and the Fermi level is located at 7.49630 ev (Akinlami et al., 2018), as shown in Figure 1B. Because the maximum value of the valence band and the minimum value of the conduction band are located in k-space at the same position (Akinlami et al., 2018), this means that the material is a direct bandgap semiconductor, and it is also a narrow bandgap semiconductor (Zhai et al., 2018), because its ideal forbidden band width is 0.32 ev (Tan et al., 2015). This band structure is conducive to electron transmission and gives PbTe a highly complex Seebeck coefficient (Zhai et al., 2018). In addition, due to the strong non-harmonics caused by the local eccentricity of Pb, PbTe has a lower lattice thermal conductivity (2.2 W/mk) (Zhu et al., 2020). Therefore, PbTe is a promising medium temperature power generation material and it is one of the most advanced traditional TE materials that have been used in the medium temperature range for the past 50 years. In recent decades, the p-type PbTe has excellent performance, and the maximum value of zT can reach about 2.5 and the average of zT is about 1.4. However, the maximum value of zT for n-type PbTe is only about 1.8 and the average value of zT is about 1 (Luo et al., 2019), as shown in Figures 1C,D. The key to the promotion and application of this promising power generation technology is to improve the low conversion efficiency of current TE materials, which is technically evaluated by the dimensionless thermoelectric  $zT = S^2 \sigma T/\kappa$ , Where S,  $\sigma$ ,  $\kappa$ and T are Seebeck coefficient, electrical conductivity, thermal conductivity and absolute temperature, respectively (Tan et al., 2016). In this review, we discussed the strategy of improving the thermoelectric performance of PbTe by introducing foreign atom doping to increase the carrier concentration and



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introducing nanostructures to reduce the thermal conductivity to optimize the TE figure of merit. The challenges of PbTe-based TE materials and their perspectives are summarized.

# STRATEGIES TO OPTIMIZE THERMOELECTRIC PERFORMANCE

The fundamental challenge of designing high *zT* TE materials comes from the strong correlation between the carrier concentration and *S*,  $\sigma$  and  $\kappa$ , which can be improved by controlling the doping level to adjust the carrier concentration (Zhou et al., 2018). In addition, the introduction of nanostructures into the system to reduce the thermal conductivity of materials by enhancing phonon scattering is also the main method to optimize thermoelectric performance (Chen et al., 2018). Below we will briefly describe these two strategies.

# Modulate Doping to Improve Thermoelectric Figure of Merit

The chemical bond in PbTe is a mixed ion-covalent bond, so it is generally classified as a polar semiconductor. N-type or p-type

semiconductor materials can be formed by doping, where n-type semiconductor means that the concentration of free electrons is much higher than the concentration of holes and vice versa, it is a p-type semiconductor. When the dopant is not present, the atoms in the crystalline material are arranged regularly and orderly. When the dopant is present, as the temperature changes, the atoms are arranged disorderly and produce lattice defects. The different atomic sizes of various dopants cause atomic-level lattice distortion and significant strain around the impurity atoms (Cojocaru et al., 2017). It changes the material of carrier concentration, band structure and lattice compatibility, reduce the thermal conductivity (Chang et al., 2020). For PbTe-based TE materials, it is usually used to adjust donor and acceptor doping to obtain n-type or p-type TE materials, respectively. When introducing heterovalent point defects to electronically dope the material, these defects are first thought to provide carriers for the system, rather than causing charged vacancies. In order to obtain p-type PbTe, alkali metals can be used as acceptor dopants. In the experiment, most of the p-type PbTe materials with excellent performance are doped with Na and K. One of the solubility of doping atoms depends on the temperature. By adjusting the temperature, the solubility of the doping atoms can be improved. The number of holes introduced by each doping atom is as close as possible to 100%, which improves the doping efficiency (Jood et al., 2020). For example, at high temperatures, Na can enter the PbTe lattice beyond the solubility limit, converge the energy band, as shown in Figure 1E, enhance TE transmission and increase Seebeck coefficient, thereby improving thermoelectric performance (Jood et al., 2020). In recent years of research, PbTe is alloyed with elements such as Eu (Chen et al., 2017), Sr and Mn to converge the energy band and then the dislocation density is increased by doping with heterovalent atoms (Jood et al., 2020). As the concentration of heterovalent atoms further increased, nano-scale precipitation appeared and the dislocation density gradually decreased. This transition from point defects to lattice defects and finally into nano-precipitation directly leads to a decrease in thermal conductivity, which further improves the zT value. For example, when the temperature is 823k, Chen et al. making a new type of alloy material EuTe is used in the energy band engineering PbTe material. Compared with Pb, Eu has a slightly larger size, which expands the crystal lattice. Eu replacing Pb can increase the unit cell size of PbTe and alloying with EuTe can promote band convergence. The solubility of Na in PbTe increases with the increase of EuTe content. When the Na content increases to a certain value, the zT value can reach 2.4 (Chen et al., 2017).

The n-type PbTe TE material can be realized by doping Ga, In, La, Sb, Al, Bi and other atoms at the Pb site and Cl, Br, I and other atoms doping at the Te site. For example, by changing the stoichiometric ratio of PbTe and doping with I atoms, various thin film samples with electron concentrations ranging from  $n = 8 \times 10^{16}$ – $2 \times 10^{18}$  cm<sup>-3</sup> can be obtained. Its Seebeck coefficient, lattice thermal conductivity, quality factor and other TE properties increase with the increase of electron concentration (Dzundza et al., 2020). Due to the intensity of the Pb atoms eccentric vibration, the band gap of PbTe increases significantly

with the increase of temperature. As the temperature rises, the band gap evolves and the band shape begins to flatten, which means that the effective carrier mass also increases (Xiao et al., 2018). Because larger effective carrier mass requires higher carrier density to achieve the best power coefficient (Su et al., 2018), it is necessary to adjust the carrier density dynamically with the increase of temperature in PbTe based materials (Chen et al., 2019). In Pb<sub>0.98</sub>In<sub>0.005</sub>Sb<sub>0.015</sub>Te, the impurity level formed by adding In element will dynamically optimize the carrier concentration in the entire working temperature range with the increase of temperature (Zhang et al., 2018), as shown in Figures 1F-H. This is because when In is doped in PbTe, its deep defect state can trap electrons, so that the trapped electrons at low temperatures are thermally activated to return to the conduction band at high temperatures. As the temperature increases, the charge state of In changes from a mixed charge state to a complete +3 state, so the carrier concentration can be increased (Bali et al., 2016). Increase the carrier density from  $2.18 \times 10^{19}$  cm<sup>-3</sup> at 300 K to  $4.84 \times 10^{19}$  cm<sup>-3</sup> at 823 K and further adjust the conductivity, Seebeck coefficient and electrical transmission properties. Then, the phonon propagation is suppressed by alloying with sulfur to form a complete solid solution and at the same time, it is beneficial to the carrier mobility to maintain high power factor, as shown in Figure 1I. The lattice thermal conductivity can be further reduced from  $0.76 \text{ Wm}^{-1} \text{ K}^{-1}$  to  $0.42 \text{ Wm}^{-1} \text{K}^{-1}$ . Combining the advantages of doping and S alloying,  $zT_{max} = 1.4$ and  $zT_{ave} = 0.87$  (300–873 K) can be obtained in Pb<sub>0.98</sub>In<sub>0.005</sub>Sb<sub>0.015</sub>Te<sub>0.94</sub>S<sub>0.06</sub> (Wang et al., 2019).

### Manipulate on Phonon to Reduce Lattice Thermal Conductivity

One of the important factors to achieve high TE conversion efficiency is the need for low thermal conductivity. The lattice thermal conductivity  $\kappa_1$  can be described as:  $\kappa_1 = \int \kappa_s(f) df$ , Where  $k_s(f)$  is the thermal conductivity of the spectral lattice, which is proportional to the square of the phonon velocity, so we can choose to manipulate the phonons to suppress the lattice thermal conductivity (Fan et al., 2010). Phonon transmission and diffusion in crystalline materials is closely related to the mean free path, and lattice defects can strengthen the phonon scattering of different wavelengths. Because the microstructure can simultaneously distort the crystal lattice and provide highdensity phase boundaries in the matrix, it has the effect of reducing thermal conductivity. Precisely controlling the size of microstructures (especially nanoinclusions) can effectively scatter phonons (Shi et al., 2020). In PbTe, its mean free path dominates on the nanometer scale, and semiconductor nanostructures have many unique advantages (Chen et al., 2015), so the use of nanostructures to reduce the thermal conductivity of PbTe lattice is a promising method. By designing special microstructures (multi-scale microstructures including point defects, line defects, interface defects, and volume defects) inside PbTe-based TE materials, enhancing electron scattering and inhibiting lattice thermal conductivity to obtain high zTmaterials (Liu et al., 2018). Since the conduction band is not as complex as its valence band, it is more necessary to design the



microstructure in the n-type PbTe than the p-type PbTe. When the temperature is 773 K, the zT peak value of 1.8 can be obtained in the n-type PbTe-InSb with special nanostructure (Zhang et al., 2017). In the nanostructured n-type PbTe, the conductivity and heat transfer performance have been improved. The energy barrier caused by the nanostructure can filter the carriers, strengthen the Seebeck coefficient and at the same time help to produce strong phonon scattering and reduce the thermal conductivity of the lattice (Xiao and Zhao, 2018). When the defect size is equal to the average free path range, the phonon scattering is significantly enhanced, which can greatly reduce the lattice thermal conductivity and improve the TE properties of PbTe (Fu et al., 2017). On the basis of this theory, some researchers have proposed a method of using full-scale hierarchical architecture to design defects to significantly reduce the thermal conductivity of the lattice (Xiao and Zhao, 2018), as shown in Figures 2A-C. This method was used by Biswas et al. to improve the zT value of p-type PbTe-SrTe to 2.2. In the spark plasma sintered samples, due to the existence of interface potential at the grain boundary, more electrons are scattered than holes (Biswas et al., 2012). The overall decrease of lattice thermal conductivity can be attributed to the extra scattering and impedance of the mesoscale grain boundary (Chen et al., 2021). Wu et al. mixed use of modulated doping and layered structure, prepared 3% Na-doped (PbTe)<sub>1-x</sub> (PbS)<sub>x</sub> materials by plasma discharge sintering and studied their TE properties. Doping excessive Na in the PbTe-PbS matrix, the diffusion and re-dissolution of impurities in the crystal grains increase the hole concentration (Watcharatharapong et al., 2017; Sharma et al., 2020), so that the Seebeck coefficient and conductivity reach saturation at 600 K and the layering mechanism makes the nano-precipitation phonon scattering

efficiency higher and the highest zT value reaches 2.3 when the temperature is 923 K (Wu et al., 2015).

# CHALLENGES OF PBTE-BASED THERMOELECTRIC MATERIALS

PbTe is a semiconductor TE material with excellent performance in the mid-temperature region. It is widely used in deep space exploration and waste heat recovery systems, such as radioisotope thermoelectric generators (as shown in **Figure 2D**) and waste heat recovery in automotive thermoelectric generators. In addition, researchers have also found it can be used to manufacture flexible TE conversion equipment for wearable applications and many other fields (Yang et al., 2017), as shown in **Figures 2E,F**. These devices have many advantages, such as excellent reliability, no noise and no pollution.

For how to improve the thermoelectric performance of PbTebased TE materials, researchers have proposed a variety of strategies, such as nanostructures, adjustment of doping and alloy concentration, energy band engineering, design of allscale layered architectures, synergistic effects, etc. The trend of future improvement should be to maximize the performance of TE by using collaborative strategies and integrating all effective factors to the greatest extent. At the same time, the improvement of the thermoelectric performance of n-type PbTe cannot be ignored. The use of reducing point defects to obtain higher carrier mobility (Wang et al., 2019) and the design of a comprehensive layered structure to disperse the whole field phonons (Liu et al., 2019) are important ways to increase the zT value. Secondly, there is huge room for improvement in TE conversion efficiency. The

PbTe-Based Thermoelectric Materials

maximum conversion efficiency of PbTe based nanostructure module is about 8.8% at a temperature difference of 570 K (Tan et al., 2019). When  $Bi_2Te_3$  is used for segmentation, the efficiency increases to about 11% at a temperature difference of 590 K, which is a record high value of PbTe based system (Tan et al., 2019). Exploring these possibilities and further improving thermoelectric performance can make TE materials and equipment more suitable for daily life.

## CONCLUSION AND OUTLOOK

Compatible PbTe thermoelectric semiconductor materials are the key components of TE devices. For PbTe thermoelectric materials, it is a strategy to dynamically dope PbTe with elements whose solubility is closely related to temperature. Another method is to introduce a deep level defect, which provides holes as acceptor impurities at low temperatures, and at high temperatures, the deep defect level is excited to provide electrons to optimize the carrier concentration. When the dislocation density is uniform, its scattering is about the same as the phonon-phonon scattering at room temperature (Shi et al., 2020), which can reduce the thermal conductivity by about 80%. Therefore, the study of dislocation scattering in the crystal may be the main direction that can reduce the thermal conductivity. The

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use of n-type and p-type dopants has also been widely used in various materials to prepare high-performance thermoelectric devices. For example, the use of n-type and p-type doping to produce a p-n junction with a power density of 1.18 mW cm<sup>-2</sup> TE module (An et al., 2017).

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All authors listed have made a substantial, direct, and intellectual contribution to the work, and approved it for publication.

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