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Effect of tungsten contents on the jet penetration performance of shaped charge liner based copper-tungsten composites

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Herein, we investigated the effect of W content on the jet penetration performance of W-Cu shaped charge liners by using both simulation and experimental methods. The W-Cu composite liners were prepared directly by using spark plasma sintering (SPS) technique. Microstructural observations showed that W particles were uniformly dispersed within the Cu matrix. The relative density of W-Cu composites decreased slightly from 99.2% to 98.8% with an increase in the W content. The hardness of the W-Cu composite liner increased as increasing W content and reached the highest value of 209.2 HV for the composite reinforced by 60 wt.% W. Besides, the penetration depth increased and reached the maximum value of 80 mm for the composite liner containing 50 wt.% W which is improved by about 11% compared to pure Cu liner. According to simulation and experiment results, the penetration depth of the W-Cu composite liners exhibits a nearly identical trend. W-Cu composite liner containing 50 wt.% W remains the best performance compared to other composites. However, the experimental results are lower compared to the simulation results. This could be because the simulation procedure did not completely account for the actual test conditions.

KEYWORDS

W-Cu composite, penetration depth, shaped charge liner, spark plasma sintering, density, hardness

1 Introduction

Shaped charges are the explosive devices that concentrate the energy of a convex explosive block, deform the liners and apply it to the convex surface of the explosive block, forming a metal jet and then moving at extremely high speeds to penetrate steel, concrete, and stone... (Ma et al., 2016; Ma et al., 2022). Shaped charges are used in many industrial fields such as oil and gas exploitation, transportation, mining and defense sectors (Wang et al., 2019; Yi et al., 2019; Baykara et al., 2021; Sun et al., 2021; He et al., 2022; Jiang et al., 2022). A typically shaped charge consists of four main components: (1) the liner, (2) the explosive, (3) the shell, and (4) the detonator. In particular, the liners have many different shapes, depending on the purpose such as: conical (for penetrating steel),

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TABLE 1 Composition and theoretical density of the W-Cu composites with different W contents.

Samples	Cu contents (wt.%)	W contents (wt.%)	Theoretical density (g/cm ³)		
Cu	100	0	8.96		
Cu70W30	70	30	10.67		
Cu60W40	60	40	11.40		
Cu50W50	50	50	12.23		
Cu40W60	40	60	13.19		

spherical (for making wide holes), or funnel-shaped (for creating pipes). Many different materials have been tested for liners (Xu et al., 2019; Sun et al., 2023). However, copper (Cu) was the first material to be used due to its high density and good ductility deformation (W. B. Li et al., 2015). Using a liner made of gold (Au) could obtain a penetration up to 47% deeper than that of the Cu liner as testing on the same target materials. However, the cost of an Au liner is much higher than that of a bronze liner. Metals with face-centered cubic (FCC) crystal structures have better plastic deformation ability than those with body-centered cubic (BCC) crystal structures (Feng et al., 1996). Therefore, the metals such as tungsten (W), and tantalum (Ta) are not selected as liner materials because of their poor plastic deformation ability. Copper is still widely used as liner material in almost studies related to shaped charge (Ahmed and Qadeer Malik, 2017). Liu et al. (2017) combined simulation and experiment to study the structural change and thermodynamic parameters of the metal jet formed from the Cu liner. Park studied the change in the structure of the liner fabricated by Ta before and after penetrating the steel (Park et al., 2019). In addition, aluminum (Al), Titanium (Ti), Zirconium (Zr), Uranium (U), and Silver (Ag). were also mentioned as metals that have been studied and tested for liners of the shaped charges (Elshenawy and Li, 2013; Saran et al., 2013; Wang et al., 2014) used Ni and Mo metals as the shaped charge materials and concluded that the penetration of the Mo liner is deeper than that of the Cu and Ni liner by 10.0% and 21.3%, respectively (Xu et al., 2019).

In addition to pure metals, many studies have also focused on the development of alloys for liners. The W-Cu-Ni alloy was studied by He and Jia (2010), while Lee (2002) used Cu-20%Pb alloy for the liner material of the shaped charges used in the oil and gas industry. Sun et al. studied two types of alloys including Ni-Al and Cu-Ni-Al used liners for shaped charges and found that Cu-Ni-Al alloy has 42% greater penetration depth than Ni-Al alloy Sun et al. (2018), Cui et al. (2020) studied the Zr-based amorphous alloy as a liner material. The study has shown the optimal focal length for the amorphous Zr alloy material but also shows that the Zr base amorphous alloy has a smaller penetration depth than copper (Cui et al., 2020). The penetration depth of the shaped charges depends on the density of the liner and the jet length. Therefore, to increase the penetration depth, it is necessary to increase the density while maintaining the high plastic deformation capacity of the material. Materials with a high density such as Mo, W or Zr have high density but low plastic deformation ability, whereas materials such as Al, Sn have good plastic deformation capacity but low density, so they hardly improve the penetration depth. Subsequent studies focused on using Cu matrix composites reinforced by high-density materials such as W, Mo. to increase the density and at the same time maintain the good ductility of the Cu matrix. Studies using Cu-W or Cu-Mo composite liner were conducted in the late 90s of the 20th centuries (Li et al., 2015; Ahmed and Qadeer Malik, 2017). It is well-known that, W-Cu is classified as a "pseudo alloy" because the W and Cu components do





not form a solution or intermetallic compounds. W-Cu composites provide excellent thermal, electrical, and mechanical performance due to their integrated properties of high hardness and strength, a low thermal expansion coefficient, good arc resistance from W, and high electrical and thermal conductivity from Cu. This class of materials is widely used in civilian industrial fields such as high-voltage electric contact parts, welding electrodes, electronic packaging, and thermal sinks, and it is also in high demand in the aerospace industry (Zhang et al., 2023). Specially, W-Cu alloy has enormous potential in the area of shaped charge liner materials due to its high density, high velocity, and high ductility (Wang et al., 2021; Zhuo et al., 2022). Some research groups investigated the jet velocity of W-Cu and Cu liners and found that the tip velocity of the W-Cu jet was significantly higher than that of the Cu jet (Seok-Hwan, 2005; Zhang et al., 2010; Ibrahim et al., 2014; Wang et al., 2014; Guo et al., 2016; Wang et al., 2022; Chen et al., 2023; Li et al., 2024). During the jet penetration process, there was a large adiabatic temperature rise at the impact site of the W-Cu jet under ultrahigh pressure, resulting in the melting of the Cu phase. The melted Cu flowed around the W particles and formed a Cu-rich layer





between the jet and the target, and the Cu-rich layer played an important role in insulating the jet and the target (Zhang et al., 2010). The W phase with a high melting point moved easily in the melted Cu phase and made a major contribution to the penetration depth. Wang et al., fabricated Cu-W composite liner (80Cu-20W) with a small amount of nickel (0.5%) and observed that the penetration depth increased by 30% as a result of the higher density of the composite liner compared to the Cu liner (Wang and Zhu, 1996). Similarly, Zygmunt and Wilk fabricated Cu-W composite liner using electrolytic Cu powder (Zygmunt and Wilk, 2008). The study showed that, although the jet velocity was lower, the penetration depth was still greater due to the increased liner density (12.5 g/cm3). In another study conducted by Elshenawy, Cu-W composite cones were fabricated by the pressing technique from Cu-W powder mixture and compared with Cu liner (Elshenawy et al., 2018). This study also shows that the penetration depth is larger than when using a Cu liner (Elshenawy et al., 2018). Zhao et al.

(2016) reported that the addition of Zn and Ni to the W-Cu composite liner will reduce the penetration depth but increase the penetration area on the target. In addition, Zhao suggested that the addition of Zn will lead to an increase in the interface bond strength between the W particles and the Cu matrix (Zhao et al., 2016; Yan et al., 2020) fabricated a multi-layer composite liner from W-Ta and W-TiN-Ta materials and showed that the mechanical properties of the liner have been significantly improved. In addition, the shaped charge test also demonstrated that this composite has a high potential for making liners of shaped charge devices (Yan et al., 2020). General, up to now, Cu-W composite is still being used for fabricating the metal liners of the shaped charge devices due to the increase in penetration depth. Most of the studies mainly focused on improving the penetration depth through the optimization of the structure and the properties of the liner material. In generally, W-Cu composites could be prepared by using powder metallurgy (PM) technology. It is usually required to have a high sintering

TABLE 2 Dimensions of the liner, shaped charge, and steel target.										

Symbols	D1	D2	D3	D4	D5	L1	L2	L3	L4	L5	R	α
Unit (mm)	Ф31.98	Ф34.27	Ф35.4	Ф39.4	Φ80	32.4	32.5	34.5	100	22	4	59.5

TABLE 3 Some properties of Cu were used in the simulation process.

Material	Density (g/cm ³)	Heat capacity (J/kgK)	Young's modulus (GPa)	Shear modulus (Gpa)	Yield strength (MPa)	Hardness (MPa)	Melting point (K)
Cu	8.96	383	129	46	90	292	1,356

temperature and a long holding time in order to achieve the required level of densification (Zhang et al., 2023). Thus, the grain coarsening is unavoidable, and copper may leach out of the skeleton, which causes copper segregation, which in turn causes a non-homogenous microstructure and poor product performance. The most common approach used in the fabrication of W-Cu composites is known as melt infiltration. However, this technique remains some defects such as pores, copper lakes, and tungsten agglomerates, resulting in a low-quality product. Because of the significant differences in melting points, densities, and mutual solubility, W-Cu composites with ultra-fine grain are extremely challenging to fabricate. In this context, many novel technologies have been explored to enhance the properties of W-Cu composites, including, laser sintering, plasma spraying and specially spark plasma sintering, etc. (Zhang et al., 2023). Generally, SPS was used to prepare the cylinder-shaped samples for additional characterization. However, conical liners with internal apex angles ranging from 40° to 90° are the most common shape. To the best of our knowledge, no research has been done so far on the use of the SPS technique to prepare W-Cu composite in a conical shape for microstructure and performance analysis.

Therefore, this study aims to prepare directly the full density, fine grain and high performance of W-Cu composite with a conical shape for the first time by using SPS technique. The conical shaped W-Cu composites were used for shaped charge liner and tested the penetration performance on 40Cr steel target. The influence of W contents on the performance of the W-Cu composite liner by using both simulation and experimental methods. The simulation study was performed by using Ansys AutoDyn -2D software. The microstructure, hardness, and performance of the W-Cu composite liner were investigated and presented.

2 Materials and methods

2.1 Materials

A commercial Cu powder with a dendritic shape supplied by Xilong Scientific Co., Ltd., has a purity of 99.5% and a particle size in the range of 44–74 μ m (Figure 1A). The W powder with an angular shape has a purity of 99.9% and an average particle size of 50 μ m (Figure 1B).



2.2 Fabrication of conical W-Cu composite liners

The conical W-Cu composite liners with different W contents were fabricated by the powder metallurgy method (Table 1). First, W and Cu powders with the designed weight ratio along with 1% paraffin added as a binder and 10% n-hexane medium were mixed and milled using high energy ball (HEB) milling technique. The powder was thoroughly mixed and milled for 48 h with a speed of 250 rpm using W-Co hard alloy balls with a ratio of balls/powder is 2/1 to obtain W-Cu composite powder. After that, the obtained powder was taken out and dried in a vacuum oven. The obtained W-Cu composite powders were pressed in a conical mold (Figures 2A, B) at a pressing pressure of 6 MPa following mixing. After pressing and shaping, the billet undergoes a presintering process at a temperature of 930°C for 120 min in a hydrogen atmosphere to obtained the required strength before the SPS sintering stage. After pre-sintering, samples were cleansed and placed in graphite molds for SPS sintering using the SPS Labox 350 system (Sinterland, Japan). The SPS sintering procedure was conducted at 900°C for 15 min at a pressure of 32.36 MPa under a vacuum condition of 6 Pa (Figure 2C) to obtain the W-Cu liner specimen (Figure 2D).



(E,F) Cu60W40, (G,H) Cu50W50 and (I,J) Cu60W40.

2.3 Characterization

To study the structure of the cone material, the sample is taken from the center position of the liner by wire cutting method as shown in Figure 3. Then the sample was ground, impregnated with FeCl₃ solution (5 g) + HCl (10 mL) + H₂O (100 mL). The microstructure of the materials was studied using optical microscopes (Axiovert 40 MAT, Germany), scanning electron microscope (SEM, Hitachi S4800, Japan), and transmission electron microscope HR-TEM (JEOL JEM) 2100, Japan). The energy dispersive X-ray spectroscopy (EDS, Hitachi S4800, Japan) was used to analyze the composition of the composites. X-Ray diffraction (XRD) patterns of the samples were recorded by using Bruker D8 Advance X-Ray diffractometer.

The density of the samples was measured by the Archimedes method on the Japanese AND GR-202 instrument by Formula 1:

$$\gamma = \frac{m_{air}}{m_{air} - m_{water}} \tag{1}$$

where γ is the density of sample (g/cm³), m_{air} and m_{water} are the mass of the samples measured in air and water, respectively.





A microhardness tester (IndentaMet 1106, Buehler United States) was used to determine the microhardness of the samples. Five measurements of the samples were taken at various points on the sample surface to evaluate the accurate hardness results. The value of hardness is the average value of 5 measurements with an error of 2%.

To evaluate the penetration ability of the samples, a test system was set up as shown in Figure 4. The explosive device consists of a steel body, a W-Cu liner and 38 ± 1 g of H11 explosive. Steel target made of 40Cr steel (C = 0.38%-0.44%, Cr = 0.8%-1.1%) has a diameter of 80 mm and a thickness of 120 mm. The distance

from the bottom surface of the liners to the top surface of the steel target is 22 mm. The steel target and the explosive device are fixed by the screw behind the steel target, ensuring that the explosive quantity and the steel target cannot be moved during the preparation test. When detonated, the explosive block will create a shock wave, deforming the W-Cu liner into a solid stream of metal, moving through the steel target at high speed and then creating a hole. After each explosion, the steel target is taken out, and cut in half lengthwise at the center of the hole to observe and measure the penetration depth. The reported penetration depth is the average value of the two measurements.





3 Results

3.1 Simulation study

The simulation process was performed by using Ansys AutoDyn -2D software. AUTODYN is an analysis module based on the finite element method integrated into ANSYS Workbench, specialized for solving nonlinear dynamics problems for solid, flow, or gas models. In this study, the shaped charge has a structure consisting of three main elements including shell, explosive, and liner. The penetration depth (P) of the shaped charge will depend on the jet length formed after the explosive reaction occurs and the density of the liner according to Formula 2:

$$P \sim L \left(\lambda . \rho_j / \rho_t \right)^{1/2} \tag{2}$$

Where *L* is the jet length, ρ_j is the density of the liner, ρ_t is the density of the target material, and λ is the coefficient related to the jet length with a value in the range from 1 to 2. The *P* will depend



on the jet length L. In this simulation, the structure of the liner, shape charge, and steel target are shown in Figure 5. The structure of the shape charge has 03 main elements: shell (1); explosives (2), and liner (3). The shell is cylindrical, designed from 40Cr steel with an outer diameter D, and length L. The explosive block A-IX-1 has a cylindrical shape. The design dimensions for the liners and shell are presented in Table 2. The composition of the liners with specific density values according to theoretical calculations used in the simulation is shown in Table 1 and Table 3.

The simulation output will give the results of the jet length (L), the penetration depth (P) on 40Cr steel, and the jet

velocity (v_1). A-IX-1 an explosive with similar properties to H11 explosive was used for the simulation process. The A-IX-1 has a composition (93.5–95)% RDX and (4–6.5)% tamed, with a density in the range of 1.6–1.68 g/cm³ and the explosion speed was determined to be 7,500–8,000 m/s. As for the Cu liner, the simulation results show that the v_1 achieved before hitting the target was determined at about 6,357 m/s with, the L of 43 mm and the P of 80 mm. When simulating the Cu-W composite liners with different W content, the effect of liner density on the formation of the jet length and the maximum penetration depth on the steel target was evaluated. Figure 6 shows the jet

velocity of the W-Cu composite liner with different W contents. The obtained results indicated that the jet velocity of W-Cu composite liners was decreased with the increase of W content. With Cu70W30 composite, the jet velocity was calculated to be 6,288 m/s. When increasing the W content up to 60%, the jet velocity is only 5,446 m/s which is decreased by 14.3% compared to pure Cu liner.

Figure 7 shows the simulation results of the W-Cu composite liner with different W contents. Figure 8 shows the dependence of the jet length and the penetration depth of the W-Cu composite liners. The jet length of the liners tends to decrease from 43 to 40 mm when increasing W content from 0% to 60%, increasing the penetration depth. The penetration depth of the liners increased from 80 to 95 mm when increasing the W content from 0% to 50%. However, when continuously increasing the W content up to 60%, the penetration depth of the liner decreases rapidly to 85 mm, which is about 10.5% reduction compared to the composite containing 50% W content. From the simulation results, it is shown that increasing the W content will increase the density of the liners but decrease the jet velocity. The penetration depth tends to increase with increasing W content from 0% to 50% and then rapidly decreases when W content increases up to 60%. From Eq. 2, it is shown that the penetration depth depends on the jet length and the density of the liners. However, the penetration depth significantly decreased when the W content increased to 60%. This demonstrated that, in addition to the dependence on the density of the liners and the jet length generated after the explosive reaction, the penetration depth may also depend on the jet velocity. The previous computational simulation studies also indicated the dependence of the penetration depth on the distance between the shaped charge and the target (H), the density of the liner (ρ_i), target density (ρ_t), the jet tip velocity (ν_{tip}), the jet velocity(v_i) as seen in Eq. 3 with $x = \rho_i / \rho_t$ and the α is the coordinate that can be calculated by numerical simulation (Guo et al., 2019).

$$P \sim (H - \alpha) \left[\left(\frac{v_{lip}}{v_j} \right)^{\sqrt{x}} - 1 \right]$$
(3)

The maximum penetration depth of the shaped charge was achieved with a W content of 50 wt.% according to the simulation results, possibly due to the optimization between the increase in the density and the jet velocity. The jet velocity decreases with the increase in the density of the liners. As the W content increases from 0 to 50 wt.%, the jet velocity is slightly decreased. However, the jet velocity reduced significantly with the liner containing W content up to 60 wt.%, which caused a reduction in the penetration depth. As a result, the W-Cu composite liner containing 50 wt.% W content remains the largest penetration depth.

3.2 Experimental study

To validate the simulation results, the W-Cu composite liners containing different W contents were fabricated by SPS. Figure 9 shows the morphological images of W-Cu powder after milling. As can be seen, W and Cu powders were uniformly dispersed together and Cu powders were deformed during the milling process. The Cu powders were much deformed and become significantly compared to their initial shapes. In contrary, W seem still remain their initial shape. This is attributed that Cu is much softer than that of W.

Figure 10 shows the SEM images of pure Cu and W-Cu composite containing different W contents after SPS sintering. As can be observed, the microstructure of pure Cu is uniform with gray color. In contrary, with W-Cu composite, the presence of W particles (white) was observed. These W particles were fairly uniform dispersed within the Cu matrix (gray). To confirm the composition of the samples, EDS mapping and EDS spectra of W-Cu composite containing 50 wt.% W content were measured. As shown in Figure 11, the gray zone exhibited a rich Cu element (red color) and the white zone is a rich W element (blue color). EDS points also confirmed the conclusion, the composition at point A (white color) and point B (red color) are W and Cu elements, respectively.

Figure 12 shows the chemical composition of the W-Cu composite liners. As a result, the samples contain Cu and W for W-Cu composite and Cu for pure Cu with concentrations as shown in the table associated with the figures. There is a slight difference between the Cu and W contents as designed. In addition, a small amount of oxygen with content <1 wt.% can also be analyzed. This is unavoidable due to the amount of oxygen that can be present on the surface of Cu and W particles during the powders and liners fabrication. However, with a low oxygen content of less than 1%, it can also be considered that the sintering process does not cause oxidation for W-Cu composite liners. Figure 13 shows the XRD patterns of the W-Cu composite liners. The obtained results indicated that typical peaks of Cu and W were detected for W-Cu composite and only typical Cu peaks were detected for pure Cu. The typical peaks of Cu were determined at 20 of 43.8°, 50.9° and 74.4° corresponding to (111), (200) and (220) planes, respectively (JCPDS No. 003-1018). The typical peaks of W are located at 20 of 40.3°, 58.2° and 73.2° corresponding to (110), (200) and (211) planes, respectively (JCPDS No. 04-0806). Besides, a small oxidation phase of Cu was also detected at 20 of 32.51°, 36.1° and



The XRD patterns of the W-Cu composite containing different W contents (A) pure Cu, (B) Cu70W30, (C) Cu60W40, (D) Cu50W50, (E) and Cu60W40.





48.7° corresponding to (110), (11-1) and (20-2) planes, respectively (JCPDS No. 48–1548). The obtained results are good agreement with EDS analysis with small amount of detected oxygen.

Figure 14A shows the density and relative density of W-Cu composite liners containing different W contents. As a result, the density of the W- Cu composites increased with increasing W content, from 8.89 to 12.76 g/cm^3 corresponding to W content from 0 to 60 wt.%. Compared with the theoretical density, the measured density of the composites is lower. The obtained results showed that when increasing the W content from 0% to 50%, the relative density of the composites decreases slightly from 99.2% to 98.8%.



experimental results of the W-Cu composite liners with different W contents.

However, the relative density decreased significantly to 96.7% for the composite containing 60 wt.% W content. This indicated that the increase of the W content may cause the formation of W clusters. These clusters inhibited the densification process due to generated pores among W particles. These pores cannot be filled by Cu matrix during the sintering process, resulting in decreasing the relative density of composite. Besides, this may also result from the low wettability of Cu and W (Gu and Shen, 2008; Ibrahim et al., 2014; Elsayed et al., 2015). Figure 14B shows the Vickers hardness of W-Cu composite liners containing different W contents. For the pure Cu liner, the hardness was measured to be 73.8 HV. When increasing the W content, the hardness of the W-Cu composite increases rapidly, the hardness reached 126.3 HV for the composite contained 30 wt.% W and reached the highest value of 209.2 HV for the composite reinforced by 60 wt.% W. This could be explained as follow: in W-Cu composite system, W acted as reinforcement materials and Cu acted as matrix. The increase in the hardness of the composites with the addition of W could result from W acted as the keys hindered the movements of dislocations. Furthermore, the mismatch strains may develop at the W-Cu interfaces resulted from the difference of the coefficients of thermal expansion between W and the Cu matrix. The mismatch strains will block the movement of the dislocations to enhance the hardness of the composites.

The prepared W-Cu composite liners were introduced to the shape charge test. The tests were carried out under the same test conditions including the same explosive type, explosive mass and 40Cr steel target to compare the penetration depth of the prepared composite liners. Figure 15 shows the optical image of cross-section along the penetration depth on a 40Cr steel target after testing. Figure 16 shows the comparison of the penetration depth between the simulation and experimental results of the W-Cu composite liners with different W contents. The obtained results indicated that the penetration depth increases and reaches the maximum value P of 80 mm for the composite liner containing 50 wt.% W content. After that, the penetration depth decreased to 64 mm as increasing W content up to 60 wt.%. The decrease in the penetration depth of composite containing 60 wt.% is mainly attributed to the decrease in the relative density. The decrease in the relative density of W-Cu composite implied that the pores were formed with in the composite during the preparation process. It is interesting to note that the trend of the penetration depth of the W-Cu composite liners is nearly the same between the experimental and simulation results. However, the experimental results are lower than those of the simulation results. This could be due to the actual test conditions not being fully considered in the simulation process. For example, the samples were prepared by SPS remaining relative density less than 100%, but the simulation process taken into account the sample with full density (100% relative density). Therefore, the addition material processing after SPS such as spinning or rolling techniques could be a key point for preparing full density composites and thus obtain a good match between the experimental and simulation results.

4 Conclusion

We have investigated the effect of W contents on the penetration depth of the W-Cu composite shaped charge liners. The W-Cu composite liners were prepared by SPS technique with W particles uniformly distributed in the Cu matrix. The relative density of the W-Cu composites decreased with increasing W content, from 99.2% to 96.7%. The hardness of W-Cu composite liners increases with increasing W content and reached to 209.2 HV for a composite reinforced with 60 wt.% W. The liner containing 50 wt.%

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W content exhibited the best performance in the penetration against 40Cr steel compared to other composites. The penetration depth increased by roughly 11% compared to a pure Cu liner and reached a maximum of 80 mm. The obtained results have a significant impact on the field of materials science, especially about W-Cu composites and their applications in shaped charge technology.

Data availability statement

The raw data supporting the conclusion of this article will be made available by the authors, without undue reservation.

Author contributions

NT: Formal Analysis, Investigation, Methodology, Validation, Writing-original draft. NV: Formal Analysis, Investigation, Writing-original draft. VL: Investigation, Writing-original draft. LD: Formal Analysis, Investigation, Validation, Writing-original draft. PV: Conceptualization, Writing-original draft, Writing-review and editing. TT: Conceptualization, Supervision, Writing-original draft. DD: Conceptualization, Supervision, Writing-original draft, Writing-review and editing.

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