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# Hydrogen-bonded, all-organic pebax/epigallocatechin gallate membranes for CO<sub>2</sub> separation

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This study systematically investigates the structural, thermal, mechanical, and gas separation properties of hydrogen-bond (H-bond) induced Pebax/epigallocatechin gallate (EGCG) membranes, emphasizing the role of EGCG as an H-bond inducer. Pebax<sup>®</sup> 1,657 membranes were fabricated via solution casting using a mixed solvent system of water and ethanol, with EGCG incorporated at varying concentrations (0–20 wt%) to assess its impact on membrane properties. The hydroxyl-rich structure of EGCG facilitates robust hydrogen-bonding interactions with the Pebax matrix, forming a transiently crosslinked structure. This strong interaction reduces the matrix's free volume and alters its microstructure by decreasing the crystalline domain size of polyamide (PA) and enhancing the exposure of the amorphous poly (ethylene oxide) (PEO) chains in Pebax. At an optimal EGCG loading of 5 wt%, the membranes exhibited a CO<sub>2</sub> permeability of  $60.2 \pm 1.1$  Barrer and a CO<sub>2</sub>/N<sub>2</sub> selectivity of  $49.6 \pm 0.8$ , representing a 33% increase in selectivity compared to pristine Pebax membranes. These performance enhancements are attributed to the reduction in fractional free volume (FFV) due to H-bond-induced structural modifications and the increased availability of amorphous PEO chains, which enhance CO<sub>2</sub>/N<sub>2</sub> diffusivity selectivity and solubility selectivity, respectively. Moreover, mechanical testing demonstrated that the 5 wt% EGCG-incorporated membrane maintains its mechanical integrity, preserving the tensile strength of pristine Pebax while slightly improving elongation at break. Molecular dynamics (MD) simulations of FFV and solubilities corroborate the experimental observations, offering insights into the mechanisms underlying the improved gas separation performance. The results highlight EGCG as an effective H-bond inducer for tuning the properties of Pebax membranes, achieving an optimal balance between mechanical stability and gas separation efficiency at 5 wt% loading. This study provides a foundation for scaling up all-organic Pebax/EGCG membranes into high-performance membrane structures, presenting a promising approach for industrial CO<sub>2</sub> separation and carbon capture applications.

## KEYWORDS

hydrogen bond, Pebax, epigallocatechin gallate, gas separation membrane, molecular dynamics simulation

## 1 Introduction

The increasing concentration of CO<sub>2</sub> in the atmosphere has intensified the demand for efficient carbon capture technologies (Guiver, 2022). Polymeric membranes have gained attention as a promising solution due to their simplicity, cost-effectiveness, and low environmental footprint (Han and Ho, 2021; Lei et al., 2020). Among these, poly

(ethylene oxide) (PEO)-based membranes are particularly notable for their strong affinity with CO<sub>2</sub>, attributed to the ether oxygen in PEO chains, which enhances CO<sub>2</sub> solubility and selectivity (Liu et al., 2013; Kargari and Rezaeinia, 2020). However, the high crystallinity of high-molecular-weight PEO often introduces structural imperfections that limit its gas separation performance (Zhu et al., 2020; Kim, 2022; Kang et al., 2021). Pebax, a poly (ether-block-amide) copolymer, serves as an excellent platform for CO<sub>2</sub> separation by integrating a rigid polyamide (PA) segment for mechanical strength with a flexible poly (ethylene oxide) (PEO) segment that boosts CO<sub>2</sub> separation performance (Kim et al., 2019). However, the crystalline domains in the PA segment hinder gas diffusion, reducing overall permeability and limiting its application scope (Embaye et al., 2021). To address these challenges, researchers have explored the incorporation of organic additives and crosslinking agents, which are prized for their cost-effectiveness, processability, and consistent performance (Zou and Zhu, 2018; Knebel and Caro, 2022; Oh et al., 2025).

Various additives have been employed to improve the gas separation performance of PEO-based membranes. Porous inorganic fillers, such as metal-organic frameworks (MOFs), and porous organic fillers, including covalent organic frameworks (COFs), porous aromatic frameworks (PAFs), have shown great promise for CO<sub>2</sub> separation membranes due to their high surface area, stability, and selectivity. However, their practical application is often hindered by challenges related to complex synthesis and difficulties in large-scale polymer integration (Liu et al., 2021; Thankamony et al., 2019; Wu et al., 2021; Yan et al., 2024; Hao et al., 2014; Hou et al., 2020). Commercial polymers, such as poly (ethylene glycol), poly (acrylic acid), poly (vinyl pyrrolidone), and polytetrafluoroethylene, have also been incorporated as organic additives. These polymers offer advantages like low cost and easy processability but generally exhibit limited separation performance (Kim et al., 2024; Yu et al., 2023; Hyo Jun et al., 2024). Additionally, interaction agents have been used to disrupt the crystallization of PEO chains, enabling high selectivity and simplicity in membrane preparation. Chemically crosslinked membranes, employing agents such as thiols, acrylates, epoxies, and amines, have demonstrated excellent CO<sub>2</sub> selectivity (Kim et al., 2020; Norouzbahari and Gharibi, 2020; Shao and Chung, 2009; Li et al., 2024; Chen et al., 2023). However, these methods often require additional steps, such as heating (Shao and Chung, 2009), UV irradiation (Norouzbahari and Gharibi, 2020; Li et al., 2024), or the removal of unreacted monomers (Chen et al., 2023). In contrast, physical interaction methods based on hydrogen bonding (H-bonds) offer significant advantages due to their straightforward preparation via simple mixing. Studies have explored the incorporation of H-bonding molecules into the PEO matrix, including trehalose dihydrate (Wang et al., 2023), 4-hydroxybenzoic acid (Choi and Kang, 2016), 5-hydroxyisophthalic acid (Yoon and Kang, 2018), 2,4,6-triaminopyrimidine (Oh et al., 2025), and polyalcoholic compounds like maltitol and mannitol (Nobakht and Abedini, 2022a). These molecules effectively disrupt PEO crystallization through strong intermolecular interactions, enhancing gas separation performance.

Epigallocatechin gallate (EGCG), with its eight hydroxyl groups, emerges as a promising alternative to previously studied H-bonding molecules. It offers straightforward processing, excellent compatibility with the Pebax matrix, and improved CO<sub>2</sub>-selective

permeation through an H-bond-induced crosslinked polymer network. This study marks the first application of EGCG as an H-bonding inducer in Pebax membranes to enhance CO<sub>2</sub> separation performance. By systematically investigating the effects of EGCG on membrane structure, properties, and gas separation efficiency, this research establishes EGCG as a practical and scalable solution for advanced CO<sub>2</sub> separation technologies. Additionally, detailed molecular dynamics (MD) simulations were performed to explore the changes in gas sorption behavior induced by the incorporation of EGCG, providing deeper insights into the mechanism of performance enhancement.

## 2 Experimental

### 2.1 Materials

Pebax®1,657, composed of 40 wt% polyamide (PA) and 60 wt% polyether (PEO), was supplied by Arkema, Inc. (Paris, France). Epigallocatechin gallate (EGCG) was procured from Sigma-Aldrich. Deionized (DI) water and ethanol (EtOH) were obtained from Duksan Pure Chemicals Co., Ltd. (Korea). All solvents used were of reagent grade and employed without additional purification.

### 2.2 Fabrication of blend membranes

The blend membranes were prepared using the solvent-evaporation method. First, a polymer solution was prepared by dissolving Pebax in 5 mL of an EtOH/H<sub>2</sub>O mixture (7:3 by volume) at 70°C under vigorous stirring. Separately, various amounts of EGCG (5, 10, 15, and 20 wt% of the total solute mass) were dissolved in 3 mL of the same solvent mixture and sonicated for 30 min. The EGCG solution was then added to the Pebax solution and stirred for an additional 30 min under identical conditions. All solutions were prepared at a concentration of 5 w/v%. The resulting homogeneous mixture was cast onto a Teflon dish and left to dry at room temperature for 24 h to allow slow solvent evaporation. To ensure complete removal of the solvent, the cast membranes were further dried in a drying oven at 50°C for 24 h. The prepared membranes are designated as Pebax/EGCG X%, where X indicates the EGCG weight percentage relative to the total solute mass (5, 10, 15, and 20%, respectively).

### 2.3 Characterizations

Fourier-transform infrared spectroscopy (FT-IR; Spectrum 100, PerkinElmer, United States) was employed to analyze the chemical structures and interactions within the membranes. The crystallinity and structural properties were examined using high-resolution X-ray diffraction (XRD; Ultima IV, Rigaku, Japan) with Cu-K $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) operated at 45 kV and 200 mA. The morphologies of the membranes were investigated via field-emission scanning electron microscopy (FE-SEM; 7610F-Plus, JEOL, Japan). Thermal properties were analyzed using differential scanning calorimetry (DSC; Discovery DSC, TA Instruments) with a heating rate of 20°C min<sup>-1</sup> under a nitrogen atmosphere. Thermal

stability was assessed through thermogravimetric analysis (TGA; Discovery TGA, TA Instruments) at a heating rate of  $20^{\circ}\text{C min}^{-1}$  in air. The  $\text{CO}_2$  adsorption capacity of the membranes was measured using TGA (TGA 55, TA Instruments) at a constant temperature of  $27^{\circ}\text{C}$ . Mechanical properties, including tensile stress and strain, were evaluated using a universal testing machine (UTM; LR10KPlus Series) with a crosshead speed of  $30\text{ mm min}^{-1}$ . For consistency, all specimens were standardized to dimensions of  $2\text{ cm} \times 1\text{ cm}$ .

## 2.4 Gas separation measurement

The gas permeation properties of the dense polymer membranes were evaluated at  $35^{\circ}\text{C}$  using the time-lag method with a constant volume/variable pressure system (Airrane Co., Ltd., Korea). Each membrane had an active area of  $15.2\text{ cm}^2$ . During the measurements, the upstream pressure was maintained at 760 torr, while the downstream pressure was kept below 2 torr. Permeability was calculated by monitoring the pressure increase in the fixed downstream volume and was expressed in barrer units ( $1\text{ barrer} = 1 \times 10^{-10}\text{ cm}^3(\text{STP})\text{ cm cm}^{-2}\text{ s}^{-1}\text{ cm Hg}^{-1}$ ). The  $\text{CO}_2/\text{N}_2$  selectivity was determined by taking the ratio of the permeability values for the two gases. To ensure reliability, the average permeability and selectivity values were calculated from three replicate membrane samples.

## 2.5 Computational methods

The effect of EGCG additives on Pebax membranes was analyzed using Materials Studio 2024, employing the Forcite, Amorphous Cell, and Sorption modules. Each Pebax unit chain comprised 9 PA segments and 35 PEO segments, maintaining a weight ratio of 40:60, as previously reported (Didden et al., 2018). Four of these unit chains were combined to form a Pebax polymer chain with a total molecular weight of 10,418 g/mol. Amorphous cells for Pebax and Pebax/EGCG 5% were constructed with an initial density of  $0.1\text{ g/cm}^3$ . The Pebax cell contained five Pebax chains, while the Pebax/EGCG 5% cell included five Pebax chains and six EGCG molecules, corresponding to the weight ratio of the additives. Following cell construction, each amorphous polymer cell was equilibrated using a modified 21-step MD compression and relaxation scheme (Yeh et al., 2023) with additional two isobaric-isothermal ensemble (NPT) steps at 298 K and 1 atm for 1,000 and 5,000 ps. The COMPASSIII force field was utilized to model the molecular interactions, employing the particle-particle particle-mesh (PPPM) method for electrostatic energy summation and atom-based Van der Waals energy summation. These simulation models enabled the investigation of structural and gas sorption properties of the membranes.  $\text{CO}_2$  and  $\text{N}_2$  loading capacities were calculated using the Sorption task via the Metropolis Monte Carlo method, with atom-based summation for both electrostatic and Van der Waals interaction energies (Sun et al., 2023). The density, FFV, and  $\text{CO}_2$  uptake were calculated for different numbers of Pebax chains in an amorphous cell (Supplementary Table S2). While the density and FFV did not change significantly,  $\text{CO}_2$  uptake increased as the number of chains increased. The  $\text{CO}_2$  uptake remained relatively stable once the number of chains exceeded three,

validating the appropriate number of chains in the amorphous cell. The same MD compression and relaxation steps were performed to equilibrate the system. Subsequently, an additional NPT-MD simulation was conducted for 1,000 ns to gather data for physical analysis (Erpenbeck and Wood, 1991). To investigate the intermolecular interactions between gas molecules and functional groups in Pebax, the radial distribution functions (RDFs) of different gas-atom pairs were analyzed. RDF, denoted as  $g(r)$ , represents the probability of locating an atom at a distance  $r$  from a reference atom within an infinitesimally thin spherical shell. This probability is mathematically defined according to the method described by Hansen and McDonald (1990) (Hansen and McDonald). The RDF analysis focused on the interactions between gas molecules and functional groups such as ether oxygen in PEO and nitrogen in PA.

## 3 Results and discussion

### 3.1 Interactions and structure of membranes

First, hydrogen-bonding interactions in Pebax/EGCG membranes were investigated using FT-IR spectroscopy (Figure 1a). The FT-IR spectrum of Pebax displayed characteristic absorption bands, including a C=O stretching vibration at  $1731\text{ cm}^{-1}$ , a C—O—C stretching band at  $1,104\text{ cm}^{-1}$ , and additional bands at  $1,637\text{ cm}^{-1}$  and  $3,298\text{ cm}^{-1}$  corresponding to the stretching vibrations of H—N—C=O and N—H groups, respectively (Nobakht and Abedini, 2022b; Mohammed et al., 2019; Farashi et al., 2019). For EGCG, the band at  $3,352\text{ cm}^{-1}$  was attributed to the hydroxyl (-OH) stretching, while the band at  $1,690\text{ cm}^{-1}$  corresponded to ester C=O stretching (Pan et al., 2019; Wang et al., 2019). Upon incorporating EGCG into Pebax, a blue shift in the C=O stretching band of EGCG from  $1,690\text{ cm}^{-1}$  to  $1,709\text{ cm}^{-1}$  was observed, indicating strong intermolecular interactions between EGCG and Pebax. Deconvolution of the C=O stretching bands revealed that the secondary carbonyl band at  $1,709\text{ cm}^{-1}$  became more prominent than the primary carbonyl band of Pebax at  $1,731\text{ cm}^{-1}$  as EGCG loading increased (Supplementary Figure S1 and Figure 1b). This change in the intensity ratio ( $I_{1709}/I_{1731}$ ) highlights the strengthening hydrogen bonding interactions between EGCG and the Pebax matrix. Additional spectral shifts in the hydroxyl and amide regions further supported these interactions. Intensity changes in the N—H stretching band at  $3,298\text{ cm}^{-1}$  and the -OH stretching band at  $3,352\text{ cm}^{-1}$  confirmed hydrogen bonding between EGCG's hydroxyl groups and Pebax's amide groups. A schematic of the proposed hydrogen bonding interactions is shown in Figure 1c, illustrating the key role of EGCG in forming strong intermolecular bonds with Pebax. These hydrogen-bonding interactions play a crucial role in the structural organization of the Pebax/EGCG membranes, which is further analyzed through XRD and DSC.

XRD analysis was performed to assess the crystallinity and structural properties of the Pebax/EGCG membranes (Figure 2a; Supplementary Figure S2; Table 1). In the XRD spectrum of neat Pebax, a strong diffraction peak at  $2\theta \approx 24.04^{\circ}$  corresponds to the crystalline phase of the PA segment ( $2\theta_{\text{PA}}$ ), while a broad peak

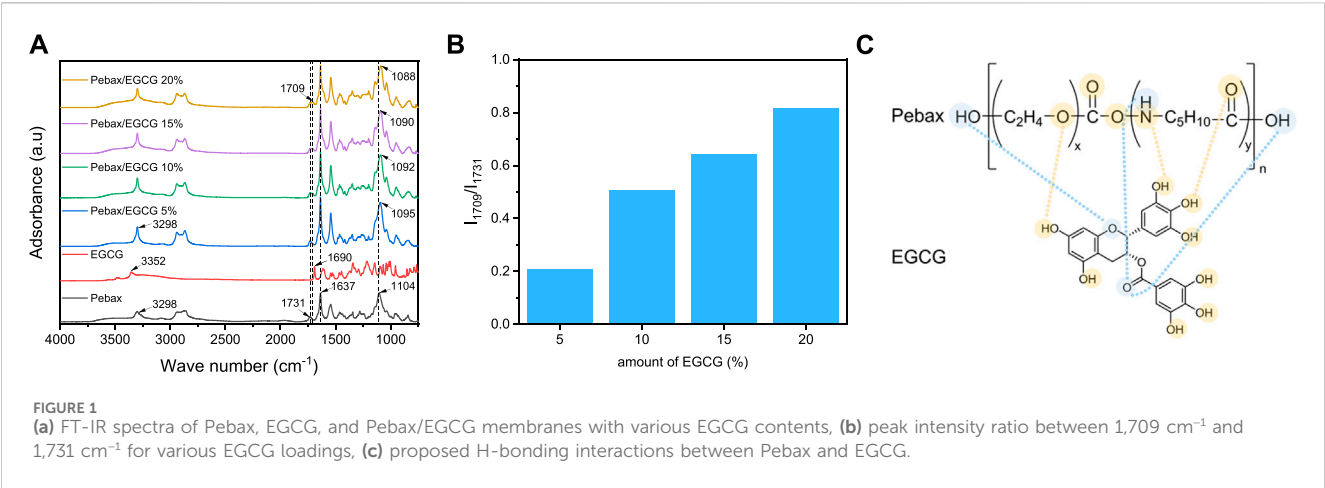


FIGURE 1 (a) FT-IR spectra of Pebax, EGCG, and Pebax/EGCG membranes with various EGCG contents, (b) peak intensity ratio between 1,709 cm<sup>-1</sup> and 1,731 cm<sup>-1</sup> for various EGCG loadings, (c) proposed H-bonding interactions between Pebax and EGCG.

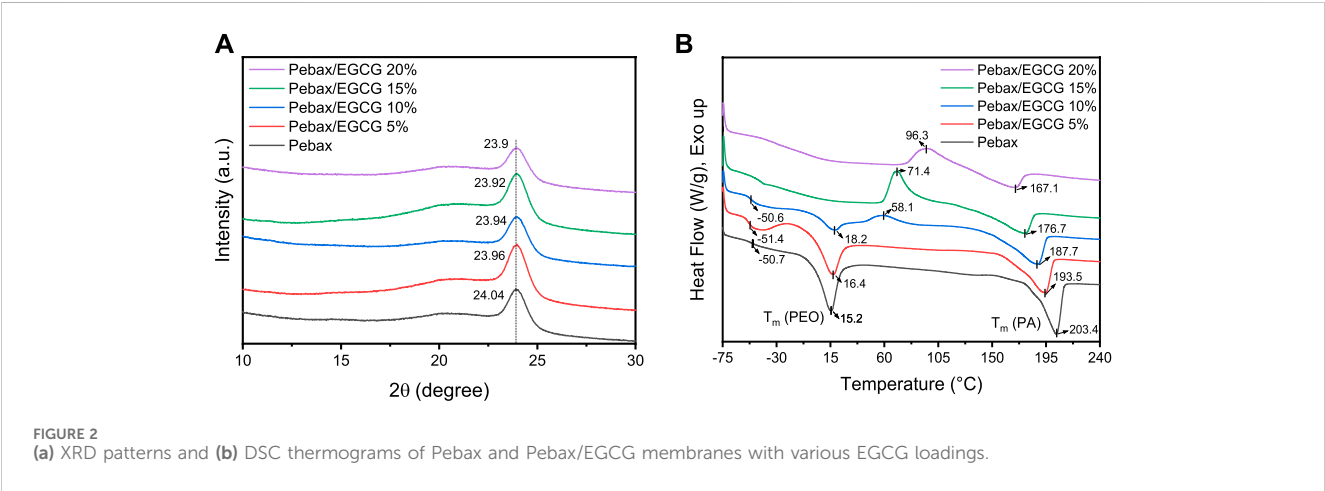


FIGURE 2 (a) XRD patterns and (b) DSC thermograms of Pebax and Pebax/EGCG membranes with various EGCG loadings.

TABLE 1 Bragg's angle of PA and PEO, intensity-based crystallinity (%), and size of the crystalline domain of Pebax and Pebax/EGCG membranes with various EGCG contents.

	$2\theta_{\text{PA}}$ (degree)	$2\theta_{\text{PEO}}$ (degree)	$I_{\text{PA}}/I_{\text{PA}}+I_{\text{PEO}}$ (%)	Size of crystalline domain, $\tau$ (nm)
Pebax	24.04	21.38	69	7.2
Pebax/EGCG 5%	23.96	21.49	67	7.0
Pebax/EGCG 10%	23.94	21.40	64	6.9
Pebax/EGCG 15%	23.92	21.51	64	7.1
Pebax/EGCG 20%	23.90	21.48	64	7.0

around  $2\theta \approx 21.38^\circ$  is associated with the amorphous phase of the PEO segment ( $2\theta_{\text{PEO}}$ ). The corresponding d-spacing values, calculated using Bragg's law, are approximately 3.7 Å for the PA segment and 4.15 Å for the PEO segment. With the incorporation of EGCG, the position of the  $2\theta_{\text{PA}}$  peak remained nearly unchanged regardless of the EGCG content.

To further investigate the impact of EGCG on the membrane crystallinity, XRD spectra of both Pebax and Pebax/EGCG membranes were deconvoluted (Supplementary Figure S2). The

intensity-based crystallinity was calculated using the following equation:

$$\frac{I_{\text{PA}}}{I_{\text{PA}}+I_{\text{PEO}}}$$

where  $I_{\text{PA}}$  is the peak intensity of the diffraction peak of the crystalline PA segment, and  $I_{\text{PEO}}$  corresponds to the amorphous PEO region. Additionally, the size of the PA crystalline domain ( $\tau$ ) was determined using the Scherrer equation:

TABLE 2 Melting temperature, crystallization temperature and degree of crystallinity of Pebax and Pebax/EGCG membranes with different EGCG loadings.

	$T_g$ (°C) (PEO)	$T_m$ (°C) (PEO)	$T_m$ (°C) (PA)	$T_{cc}$ (°C) (PA)	$\Delta H_m$		$X_{c,PEO}$ (%)	$X_{c,PA}$ (%)
					PEO	PA		
Pebax	−50.7	15.2	203.4	—	26.0	28.3	26.0	30.8
Pebax/EGCG 5%	−51.4	16.4	193.5	—	22.2	24.2	23.5	27.6
Pebax/EGCG 10%	−50.6	18.2	187.7	58.1	11.0	24.8	12.3	30.0
Pebax/EGCG 15%	—	—	176.7	71.37		20.8	—	26.5
Pebax/EGCG 20%	—	—	167.1	96.28		13.9	—	18.9

$$\tau = \frac{K\lambda}{\beta \cos \theta}$$

where  $\tau$  is the crystalline domain size,  $K$  is the dimensionless shape factor (typically 0.9),  $\lambda$  is the wavelength,  $\beta$  is the full-width at half-maximum (FWHM) of the diffraction peak in radians, and  $\theta$  is the Bragg angle.

As shown in Table 1, the intensity-based crystallinity decreased from 69% to 64% (10% EGCG incorporation) but remained largely unchanged with EGCG incorporation up to 20%. This change in crystallinity indicates that the PA chain packing is minimally disrupted at low EGCG loadings, an essential factor for maintaining the membrane's ductility and structural integrity. Additionally, the incorporation of EGCG slightly reduced the size of the PA crystalline domains ( $\tau$ ). This reduction suggests that the outer regions of the PA crystalline domains are disrupted due to hydrogen-bonding interactions between EGCG and the polymer matrix. These findings highlight that at low concentrations, EGCG acts as a hydrogen-bonding inducer without compromising the crystallinity or structural integrity of the Pebax membranes, ensuring their mechanical robustness and performance stability.

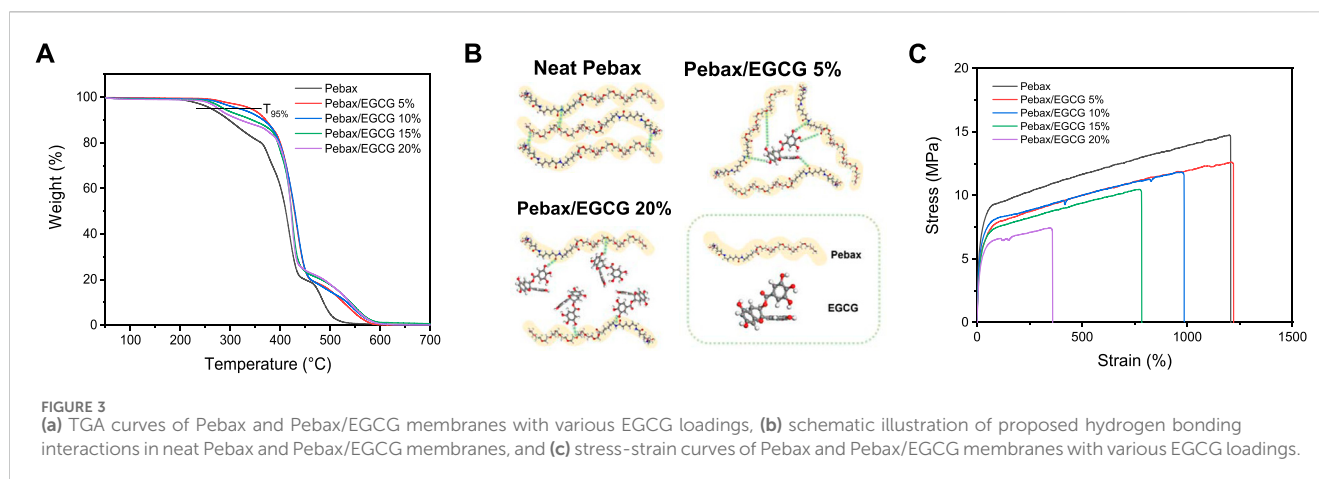
Differential Scanning Calorimetry (DSC) analysis was performed to examine the thermal transitions, including the glass transition temperature ( $T_g$ ), cold crystallization temperature ( $T_{cc}$ ), and melting temperature ( $T_m$ ) of Pebax/EGCG membranes (Figure 2b; Table 2). The DSC samples were initially heated in a nitrogen atmosphere at a rate of 20°C/min from −75°C to 250°C, then rapidly cooled back to −75°C. Subsequently, the samples were reheated to 250°C at the same rate, and the thermal transitions were analyzed using the second heating cycle. For neat Pebax, a single  $T_g$  was observed at −50.7°C, corresponding to the soft PEO phase, with no detectable  $T_g$  for the PA phase due to its low chain mobility. Up to 10% EGCG loading, the  $T_g$  of the PEO phase remained almost unchanged, indicating that low EGCG content does not significantly affect PEO chain mobility. The neat Pebax membrane exhibited strong endothermic peaks at 15.2°C and 203.4°C, representing the  $T_m$  of the soft PEO and hard PA phases, respectively, consistent with the microphase-separated morphology of the block copolymer (Thanakkasaranee et al., 2018). When 5% and 10% EGCG were incorporated, the  $T_m$  of the PEO phase increased to 16.4°C and 18.2°C, respectively, suggesting that favorable H-bond-induced interaction between EGCG and PEO chains restrict the mobility of chains, thereby enhancing the thermal stability of PEO domain. This trend (distinct  $T_m$  increase in PEO) is consistent with similar study on Pebax/polyalcohol blends (Nobakht and Abedini, 2022a).

Conversely, the  $T_m$  of the PA phase decreased to 193.5°C and 187.7°C, suggesting that hydrogen bonding between EGCG and PA chains was weaker compared to its interaction with PEO. Moreover, the decrease in the  $T_m$  of the PA phase with EGCG incorporation can be attributed to the reduction in the PA crystalline domain size ( $\tau$ ), as highlighted in the XRD analysis. At higher EGCG loadings, the  $T_m$  of the PEO phase disappeared, indicating that the PEO segments became fully amorphous due to disruption of chain packing by the increased EGCG content. Meanwhile, the  $T_m$  for the PA phase remained, further supporting the preferential interaction of EGCG with the PEO segments.

Incomplete crystallization during cooling can lead to cold crystallization upon reheating ( $T_{cc}$ ; cold crystallization temperature) due to slow crystallization kinetics (Yin et al., 2015; Gao et al., 2021). In the second heating cycles of samples with 10%, 15%, and 20% EGCG incorporation,  $T_{cc}$  of the PA phase was observed in the range of 58°C–96°C, confirming incomplete crystallization during cooling scans. Previous studies have reported the crystallization temperature ( $T_c$ ) of PA at 189.7°C and its  $T_{cc}$  at 69.9°C (Liu et al., 2003), significantly lower than the typical  $T_c$ . As shown in Supplementary Figure S3, the cooling scans of Pebax and Pebax/EGCG reveal that  $T_{c,PA}$  of Pebax progressively decreases with increasing EGCG content, eventually disappearing in the cooling scans of Pebax/EGCG 15% and 20%. This demonstrates that H-bonding interactions between EGCG and the PA phase hinder crystallization during the cooling process, resulting in the emergence of  $T_{cc}$ . Furthermore,  $T_{cc}$  increases as EGCG content rises, reflecting that higher EGCG concentrations further slow down the crystallization kinetics, necessitating additional heat input to achieve crystallization. Consequently, EGCG significantly disrupts the crystallization behavior of the PA phase and reduces the  $T_m$  of PA segments. To further understand the structural properties of the membranes, the degree of crystallinity for the PEO ( $X_{c,PEO}$ ) and PA ( $X_{c,PA}$ ) domains was calculated using the following equation:

$$X_{c,i} = \frac{\Delta H_{m,i}}{w_i \Delta H_{m,i}^0} \times 100 \text{ (%)}$$

where  $\Delta H_{m,i}$  is the heat of melting of the microphase crystals evaluated by the integrated areas under each melting peak,  $w_i$  is the weight fraction of each domain, and  $\Delta H_{m,i}^0$  is the heat of melting of the fully crystalline phase. The values of  $\Delta H_{m,PEO}^0$  and  $\Delta H_{m,PA}^0$  have been reported in previous studies as 166.4 and 230.0 J/g, respectively (Kim et al., 2019; Nobakht and Abedini, 2022a;



Rabiee et al., 2015). As shown in Table 2,  $X_{G,PA}$  does not significantly decrease with increasing EGCG content up to 15%, while  $X_{G,PEO}$  decreases markedly. This suggests that the hydroxyl groups of EGCG interact more favorably with PEO phase, disrupting the van der Waals (VdW) interactions between adjacent PEO chains and ultimately hindering the crystallization of the PEO chains. The combination of a relatively stable PA crystalline domain and an amorphous PEO domain resulting from EGCG incorporation is advantageous for CO<sub>2</sub> separation. The increased amorphous regions in the PEO phase can improve CO<sub>2</sub> permeability and selectivity, while the stable PA crystalline regions ensure the membrane's mechanical integrity.

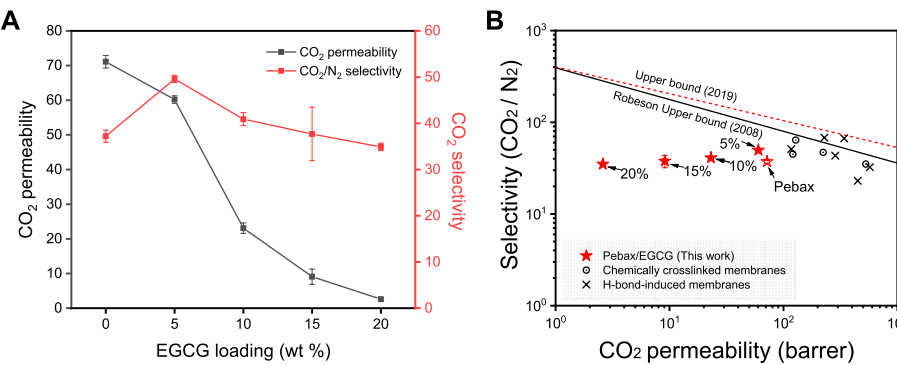
### 3.2 Thermal and mechanical properties of membranes

TGA was performed to assess the thermal stability of Pebax/EGCG membranes (Figure 3a). The TGA curves of neat Pebax exhibited a single degradation phase with a one-step decomposition profile. Upon EGCG incorporation, the onset of thermal degradation shifted to slightly higher temperatures, indicating enhanced thermal stability of the polymer matrix. The thermal degradation temperature ( $T_d$ ), or  $T_{95\%}$ , which represents the temperature at which 5% weight loss occurs, was approximately 270°C for neat Pebax. In contrast, the incorporation of EGCG raises the  $T_{95\%}$ , with the Pebax/EGCG 5% membrane demonstrating the highest thermal stability, retaining weight up to approximately 370°C. At low EGCG concentrations (e.g., 5 wt%), multiple strong hydrogen bonds formed between EGCG and the functional groups of the polymer, creating a thermally stable composite membrane. These hydrogen-bonding interactions restricted polymer chain mobility, contributing to the enhanced thermal stability of the membrane. A schematic representation in Figure 3b illustrates how EGCG molecules are uniformly distributed within the polymer matrix, forming multiple interaction sites. However, at higher EGCG loadings (e.g., 20 wt %), the excess EGCG disrupted van der Waals and hydrogen-bonding interactions among the Pebax polymer chains, leading to

reduced thermal stability compared to the optimal 5 wt% EGCG membrane. This trend highlights the critical importance of optimizing EGCG content to maintain strong interactions between the additive and the polymer matrix. Furthermore, minimal weight loss was observed for all membranes below 270°C, confirming their suitability for applications in post-combustion carbon capture, where flue gas temperatures typically range from 110°C to 140°C (Ren et al., 2012).

The mechanical properties of Pebax/EGCG membranes were evaluated using stress-strain curves (Figure 3c). Neat Pebax demonstrated remarkable flexibility, with an elongation at break exceeding 1,400%, reflecting its high ductility, and a tensile stress at break of 15 MPa. Upon incorporating 5% EGCG, the membrane's elongation at break further increased, indicating enhanced flexibility due to improved chain mobility. This observation aligns with the DSC results, which showed a decrease in  $T_g$  with 5% EGCG loading. However, the tensile stress at break slightly decreased, likely attributed to a reduction in crystallinity, as evidenced by the DSC analysis. As the EGCG concentration increased to 10%, 15%, and 20%, both the elongation at break and tensile stress at break gradually declined. This reduction suggests that excessive EGCG disrupts chain entanglement and compromises the membrane's mechanical performance. These findings emphasize the critical need for optimizing EGCG content. While low levels of EGCG enhance flexibility and maintain mechanical stability, excessive loading diminishes ductility and tensile strength, adversely affecting the overall performance of the membranes.

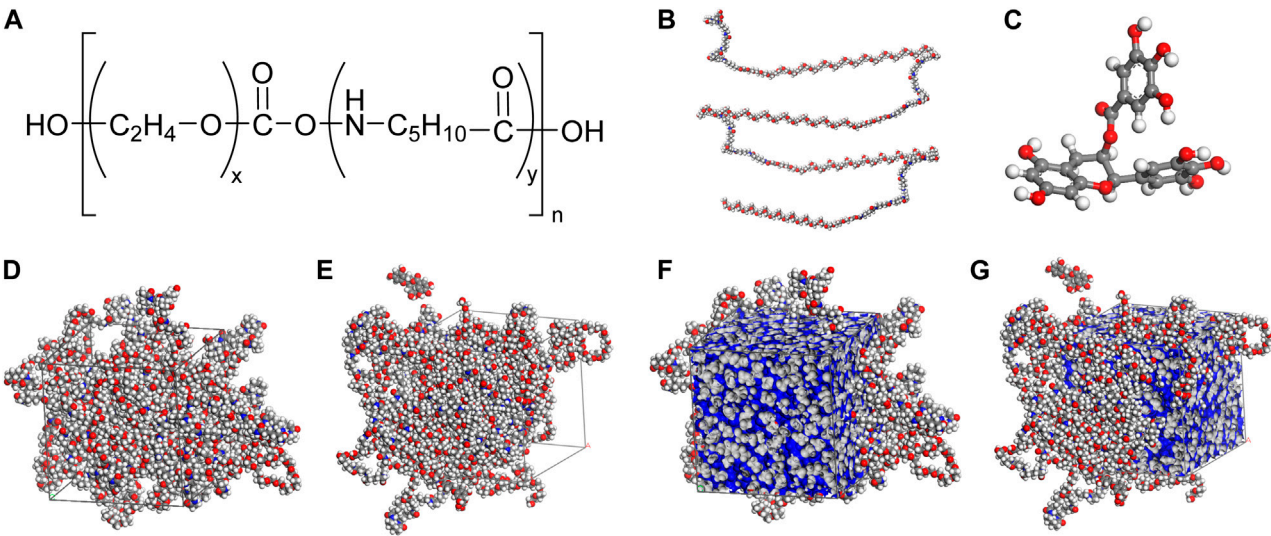
Surface and cross-sectional SEM images of neat Pebax and Pebax/EGCG membranes were analyzed to evaluate the impact of EGCG on membrane morphology (Supplementary Figure S4). The surface of the pristine Pebax membrane displayed a distinctive structure characterized by nanofibril-like PA crystalline domains embedded within amorphous PEO regions, consistent with its microphase-separated morphology (Kim et al., 2024). With the addition of EGCG, the surface morphology underwent noticeable changes, with a reduction in visible nanofibrils of PA. This suggests that EGCG interacts with and modifies the crystalline structure of PA, a finding supported by the XRD and DSC analyses. Cross-sectional SEM images of both pristine Pebax and Pebax/EGCG



**FIGURE 4** (a) CO<sub>2</sub> permeabilities and CO<sub>2</sub>/N<sub>2</sub> selectivities of Pebax and Pebax/EGCG membranes with various EGCG loadings and (b) a plot of CO<sub>2</sub> permeability *versus* CO<sub>2</sub>/N<sub>2</sub> selectivity for pristine Pebax, Pebax/EGCG membranes, and other PEO-based crosslinked or hydrogen bonded membranes previously reported in the literature.

**TABLE 3** Pure gas separation performance and thickness of Pebax, and Pebax/EGCG membranes at 1bar, 35°C.

	Thickness (μm)	CO <sub>2</sub> Permeability (barrer)	N <sub>2</sub> permeability (barrer)	Selectivity (CO <sub>2</sub> /N <sub>2</sub> )
Neat Pebax	105	71.7 ± 1.8	1.9 ± 0.1	37.2 ± 1.3
Pebax/EGCG 5%	100	60.2 ± 1.1	1.2 ± 0.04	49.6 ± 0.8
Pebax/EGCG 10%	95	23.1 ± 1.5	0.6 ± 0.09	40.9 ± 1.4
Pebax/EGCG 15%	90	9.1 ± 2.2	0.3 ± 0.08	37.7 ± 5.8
Pebax/EGCG 20%	90	2.6 ± 0.4	0.07 ± 0.01	34.9 ± 0.8



**FIGURE 5** (a) Pebax chain consisting of PEO unit (x = 35), and PA unit (y = 9) with repeating unit of 4 (n = 4). Simulated (b) Pebax chain, (c) EGCG molecule, (d) Pebax cell consisting of 5 Pebax chains, (e) Pebax/EGCG 5% cell consisting of 5 Pebax chains and 6 EGCG molecules, simulated free volume of (f) Pebax cell and (g) Pebax/EGCG 5% cell.

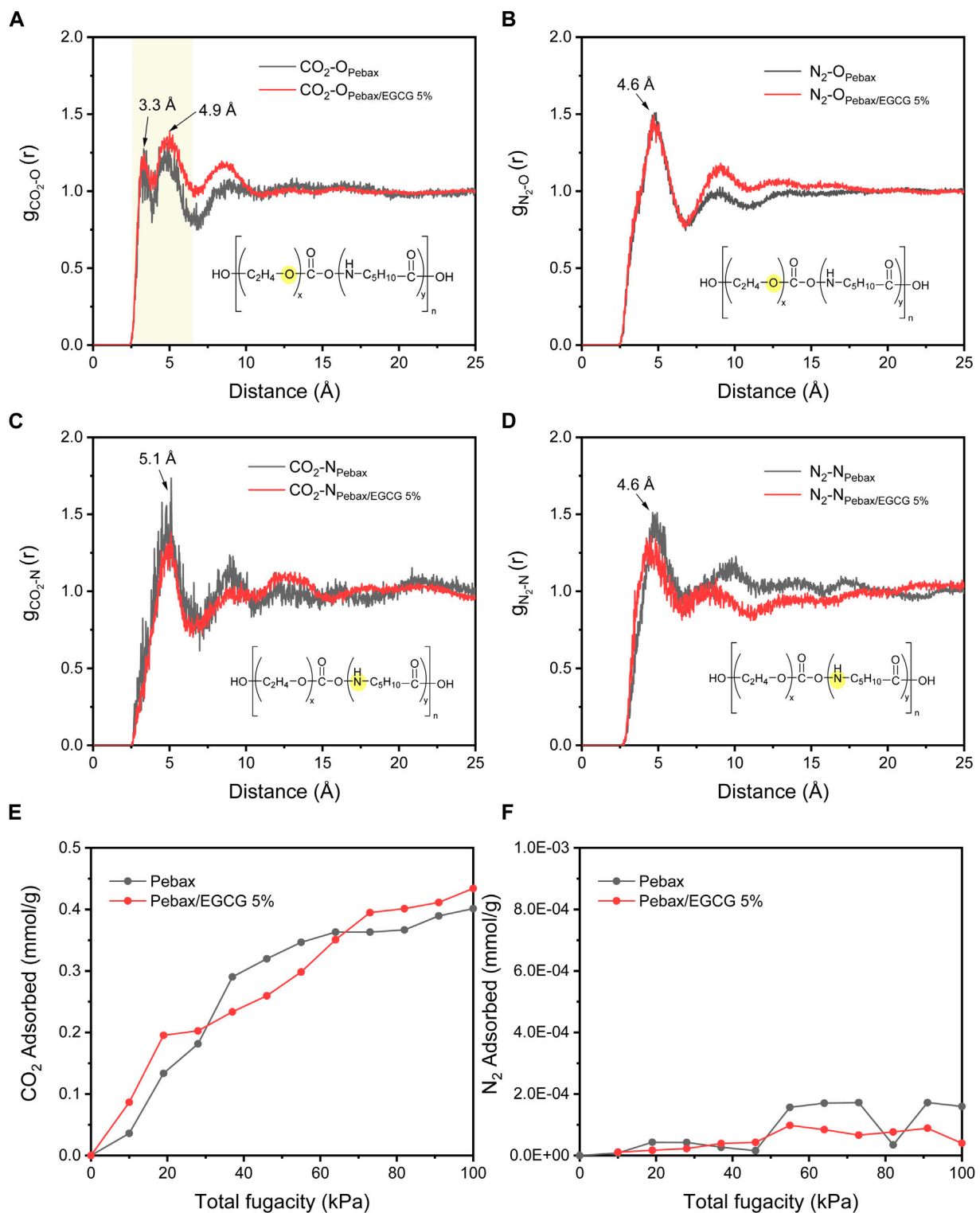


FIGURE 6  
RDF profiles of (a) CO<sub>2</sub>-O (b) N<sub>2</sub>-O, (c) CO<sub>2</sub>-N, and (d) N<sub>2</sub>-N pairs. Simulated (e) CO<sub>2</sub> and (f) N<sub>2</sub> adsorption isotherms of neat Pebax and Pebax/EGCG 5% membranes.

membranes revealed dense and uniform structures, with no visible defects. This defect-free morphology is critical for ensuring the membranes' effectiveness in gas separation applications. These

observations confirm that the incorporation of EGCG maintains the structural integrity of the membranes while influencing their microstructural characteristics.

### 3.3 Gas separation performance of membranes

The pure gas permeability and selectivity of the Pebax/EGCG membranes were measured at 35°C and 1 bar using the time-lag method, with results presented in Figure 4a and Table 3. The neat Pebax membrane exhibited a CO<sub>2</sub> permeability of 71.7 Barrer and a CO<sub>2</sub>/N<sub>2</sub> selectivity of 37.2. This performance is attributed to the structural characteristics of Pebax, where the PEO segment enhances CO<sub>2</sub> solubility, improving selectivity against non-polar gases, while the crystalline PA domain restricts gas diffusion, thereby lowering overall permeability. With increasing EGCG content, distinct trends in gas separation performance were observed. At 5 wt% EGCG loading, the CO<sub>2</sub> permeability decreased to 60.2 Barrer, while the CO<sub>2</sub>/N<sub>2</sub> selectivity increased significantly to 49.6, representing more than a 30% improvement over the neat Pebax membrane.

The enhancement in selectivity is attributed to two key factors: (1) increased CO<sub>2</sub> diffusivity selectivity due to a reduction in free volume, resulting from hydrogen-bond-induced densified structure of Pebax chains, and (2) enhanced CO<sub>2</sub> solubility selectivity caused by the greater exposure of amorphous PEO chains, facilitated by hydrogen-bond interactions with EGCG. Gravimetric CO<sub>2</sub> sorption analysis (Supplementary Figure S5) confirmed the improved CO<sub>2</sub> solubility of Pebax/EGCG 5% compared to neat Pebax. At a constant temperature of 27°C, the Pebax/EGCG 5% membrane exhibited a significantly higher CO<sub>2</sub> uptake, highlighting the role of EGCG in promoting solubility. However, higher EGCG loadings negatively impacted performance. Excessive EGCG led to increased hydrogen bonding between EGCG molecules, reducing interchain interactions within Pebax. This disrupted the selective permeation of CO<sub>2</sub> through the PEO domains, resulting in a decline in both permeability and selectivity. At 20 wt% EGCG loading, the CO<sub>2</sub> permeability dropped to 2.6 Barrer, and the CO<sub>2</sub>/N<sub>2</sub> selectivity decreased to 34.9. These results indicate that 5 wt% EGCG is the optimal loading level, where the permeability reduction caused by hydrogen bonding remains minimal, and the selectivity enhancement is maximized. In Figure 4b and Supplementary Table S1, the CO<sub>2</sub>/N<sub>2</sub> separation performance of the Pebax/EGCG membranes is compared to previously reported all-polymeric, PEO-based crosslinked or H-bonded membranes for CO<sub>2</sub> separation (Oh et al., 2025; Kim et al., 2020; Norouzbahari and Gharibi, 2020; Li et al., 2024; Chen et al., 2023; Wang et al., 2023; Choi and Kang, 2016; Yoon and Kang, 2018; Nobakht and Abedini, 2022a). Only a few of the reported hydrogen-bonded membranes exceed the 2008 upper limit (Nobakht and Abedini, 2022a).

In comparison to chemically crosslinked PEO-based membranes, such as UV crosslinked PEO (CO<sub>2</sub> permeability of 223 Barrer, CO<sub>2</sub>/N<sub>2</sub> selectivity of 47) and thiol-modified copolymer membranes (CO<sub>2</sub> permeability of 128 Barrer, CO<sub>2</sub>/N<sub>2</sub> selectivity of 64), the Pebax/EGCG membrane achieves comparable or superior selectivity while maintaining a relatively simple fabrication process without the need for chemical crosslinking. Additionally, when benchmarked against hydrogen-bond-induced membranes, such as PEO/trehalose (CO<sub>2</sub> permeability of 117 Barrer, CO<sub>2</sub>/N<sub>2</sub> selectivity of 51) and PEO/5-hydroxyisophthalic acid (CO<sub>2</sub> permeability of 573 Barrer, CO<sub>2</sub>/N<sub>2</sub> selectivity of 32.4), the Pebax/EGCG membrane demonstrates a favorable balance between permeability and selectivity. Furthermore, while polyalcohol-blended Pebax membranes (e.g., Pebax/maltitol, CO<sub>2</sub> permeability of 341 Barrer, CO<sub>2</sub>/N<sub>2</sub> selectivity of 66.65) exhibit higher

permeability, they require significantly higher operating pressures (10 bar), whereas the Pebax/EGCG membranes achieve enhanced selectivity under mild conditions (1 bar). Among both the Pebax and Pebax/EGCG membranes, the Pebax/EGCG 5% membrane demonstrates the most promising performance. This membrane will be further developed into a thin-film composite membrane to assess its scalability for practical applications.

### 3.4 MD simulation

The simulated models of Pebax, EGCG, Pebax cell, and Pebax/EGCG 5% cell are shown in Figure 5. Thermodynamic and physical equilibrium conditions of each model were confirmed (Supplementary Figure S6). The final snapshots of each simulation cell from the molecular dynamics simulations were used to calculate the fractional free volume (FFV) of the Pebax and Pebax/EGCG 5% cells. The FFV was determined using the following method (Salestan et al., 2021):

$$\text{FFV}_{\text{sim}} = \frac{V - V_0}{V}; V_0 = 1.3 \times V_{\text{vdw}}$$

where  $V$  is the total volume of the simulation cell and  $V_{\text{vdw}}$  is the van der Waals volume computed with atom volumes and surfaces task in Materials Studio 2024 using a Connolly radius of 0 Å and a van der Waals scale factor of 1. The simulated density (1.14 g/cm<sup>3</sup>) and FFV (0.138) of Pebax closely align with the experimental values (1.14 g/cm<sup>3</sup>, 0.12–0.13) (Shin et al., 2019; Sharma et al., 2019), confirming the accuracy of the chosen repeating unit. The Pebax/EGCG 5% membrane exhibited a slightly higher density (1.15) and lower FFV (0.134) compared to Pebax, which accounts for its reduced gas permeability.

Radial distribution functions (RDFs, Figures 6a–d) were used to analyze CO<sub>2</sub> and N<sub>2</sub> distribution around functional atoms in Pebax and Pebax/EGCG 5% membranes. The RDFs of CO<sub>2</sub> around PEO oxygen ( $g_{\text{CO}_2-\text{O}}(r)$ ) and PA nitrogen ( $g_{\text{CO}_2-\text{N}}(r)$ ), and N<sub>2</sub> around PEO oxygen ( $g_{\text{N}_2-\text{O}}(r)$ ) and PA nitrogen ( $g_{\text{N}_2-\text{N}}(r)$ ), were analyzed to assess changes in gas-functional group interactions in Pebax with EGCG incorporation.  $g_{\text{CO}_2-\text{O}}(r)$  exhibited a two-layer conformation with a first maximum at 3.3 Å, whereas  $g_{\text{CO}_2-\text{N}}(r)$ ,  $g_{\text{N}_2-\text{O}}(r)$ , and  $g_{\text{N}_2-\text{N}}(r)$  showed first maximums at 4.6–5.1 Å, indicating that the Coulombic interaction is only present in CO<sub>2</sub>-O (Xu et al., 2011). The stronger CO<sub>2</sub>-O bonding in Pebax/EGCG 5%, indicated by higher peak intensities in  $g_{\text{CO}_2-\text{O}}(r)$  (Xu et al., 2011), suggests enhanced CO<sub>2</sub> solubility in Pebax/EGCG 5% membrane compared to neat Pebax. Gas sorption simulations (Figures 6e, f) using the Metropolis Monte Carlo method revealed higher CO<sub>2</sub> loading and lower N<sub>2</sub> loading in Pebax/EGCG 5%, consistent with the experimental CO<sub>2</sub> sorption results (Supplementary Figure S5). This indicates enhanced solubility selectivity in Pebax/EGCG 5% compared to neat Pebax. Overall, the MD simulation results align with the experimental data, showing a reduced FFV (indicating lower diffusivity) and increased CO<sub>2</sub> solubility in Pebax/EGCG 5% relative to neat Pebax.

## 4 Conclusion

This study presents a comprehensive analysis of the structural, thermal, mechanical, and gas separation properties of all-organic,

H-bonded Pebax/EGCG membranes, highlighting their potential for CO<sub>2</sub> separation applications. Detailed physicochemical analyses demonstrated that the hydroxyl groups of EGCG effectively interact with the Pebax matrix through H-bonding, which disrupts the crystalline structure of the PA domains and exposes the amorphous PEO chains. This structural modification significantly influences the membrane's gas separation performance. MD simulations supported these findings, confirming the changes in sorption behaviors with EGCG incorporation. At low EGCG loadings, H-bond-induced crosslinking between Pebax chains and amorphous PEO segments reduces the FFV, enhancing both CO<sub>2</sub>/N<sub>2</sub> diffusivity selectivity and solubility selectivity. The membrane with an optimal EGCG loading of 5 wt% achieved a CO<sub>2</sub> permeability of 60.2 Barrer and a CO<sub>2</sub>/N<sub>2</sub> selectivity of 49.6 (133% of neat Pebax). This balance of properties demonstrates the effective tuning of the Pebax membrane structure through controlled H-bond interaction. However, excessive EGCG loading (20 wt%) led to overly strong interactions and a densified structure, causing a significant reduction in CO<sub>2</sub> permeability (3.6% of neat Pebax) and selectivity (94% of neat Pebax), highlighting the importance of optimizing the additive concentration. In conclusion, the study establishes 5 wt% EGCG as the ideal concentration for Pebax membranes, achieving a balance between flexibility, structural integrity, and enhanced gas separation performance. This work highlights the innovative use of EGCG as an H-bond inducer to fine-tune membrane properties, offering a promising strategy for developing advanced materials for CO<sub>2</sub> separation. Future work will focus on transforming the freestanding membrane into a thin-film composite membrane to evaluate its industrial viability, paving the way for practical applications in carbon capture technologies.

## Data availability statement

The original contributions presented in the study are included in the article/**Supplementary Material**, further inquiries can be directed to the corresponding author.

## Author contributions

JH: Conceptualization, Data curation, Writing–original draft. MK: Conceptualization, Data curation, Writing–original draft. NO:

Data curation, Writing–original draft. JK: Conceptualization, Supervision, Writing–review and editing.

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## Conflict of interest

The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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## Supplementary material

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/frmst.2025.1541236/full#supplementary-material>

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